

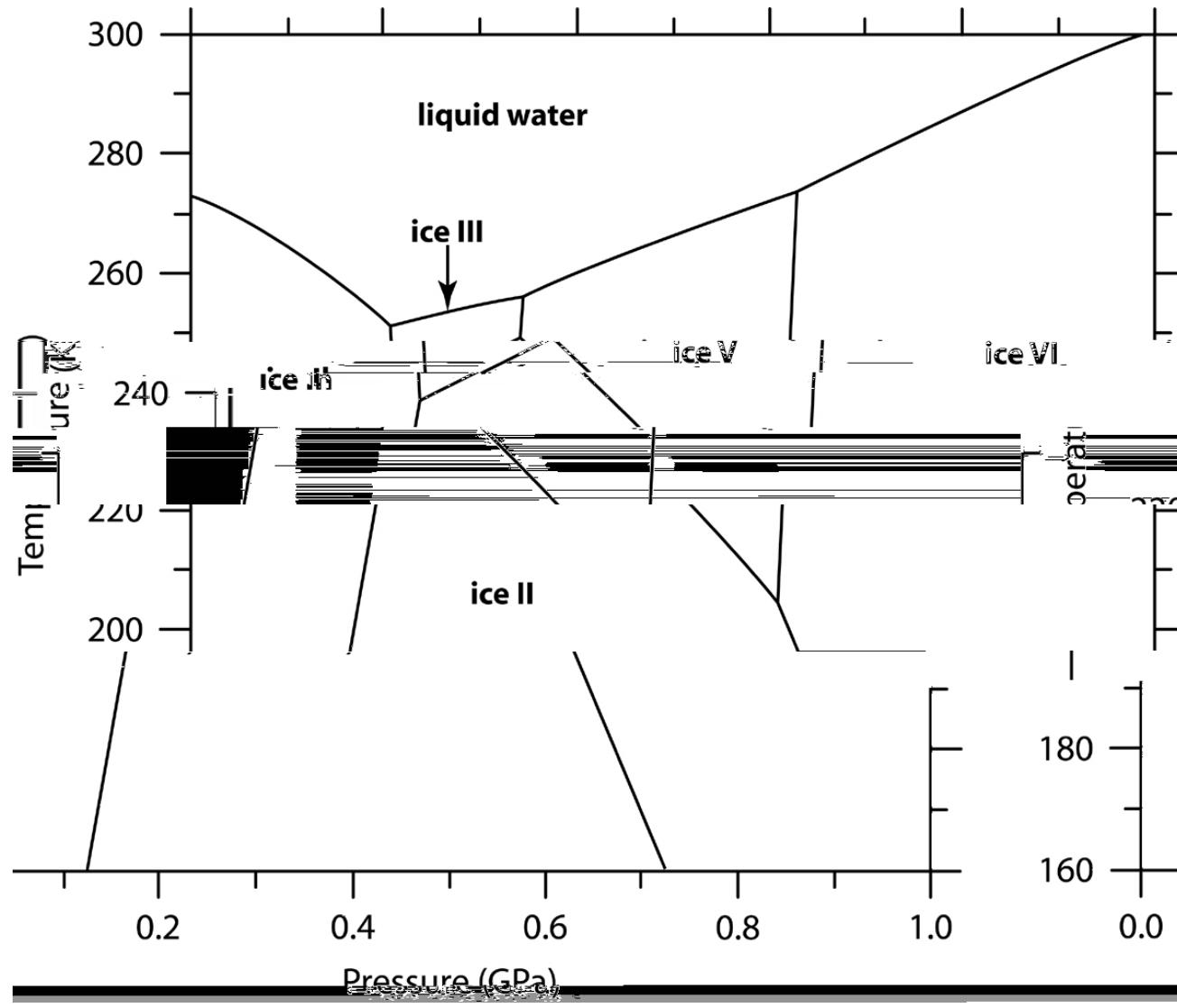
# Phase diagrams from ab-initio simulations

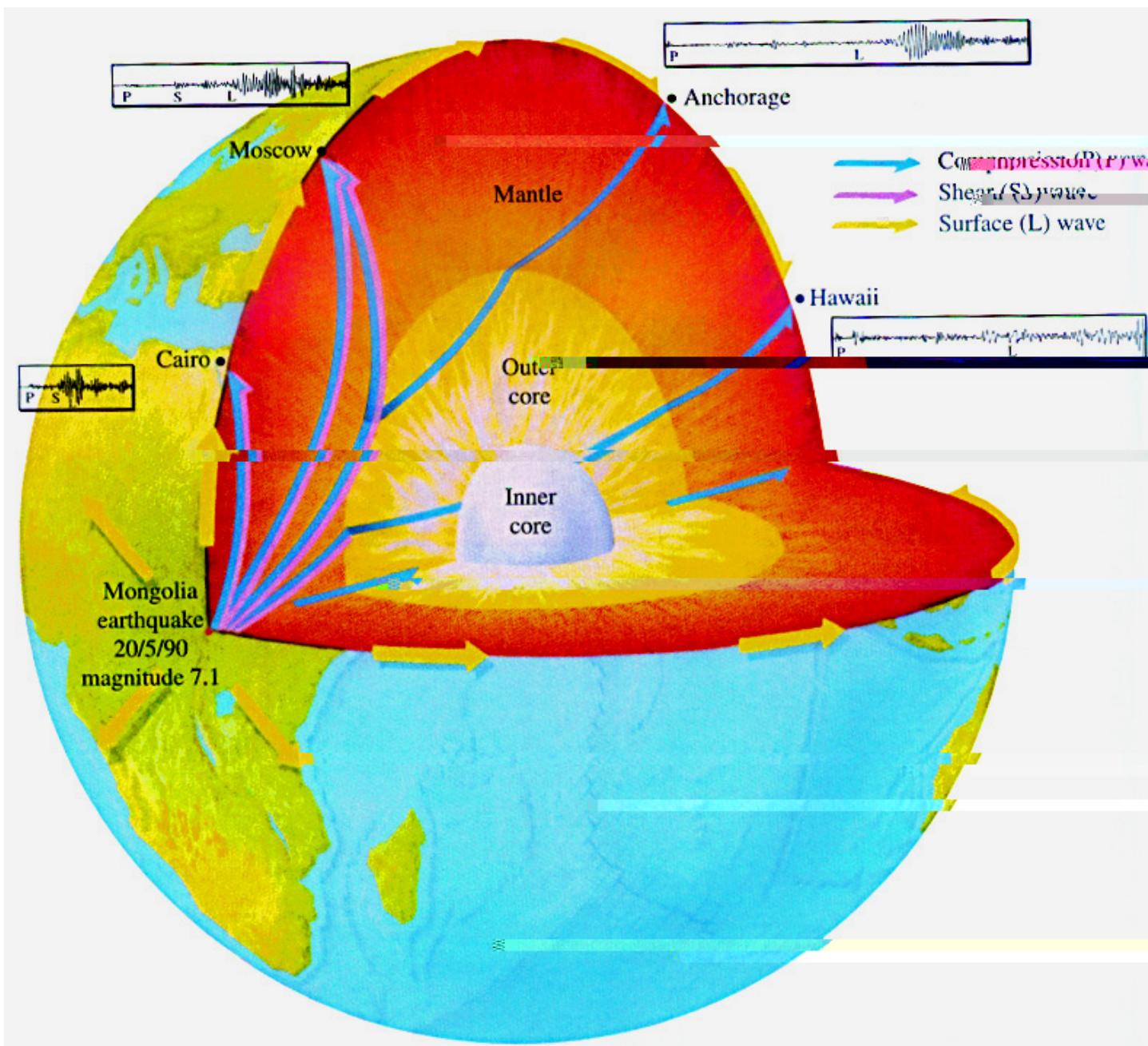
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*University College London*

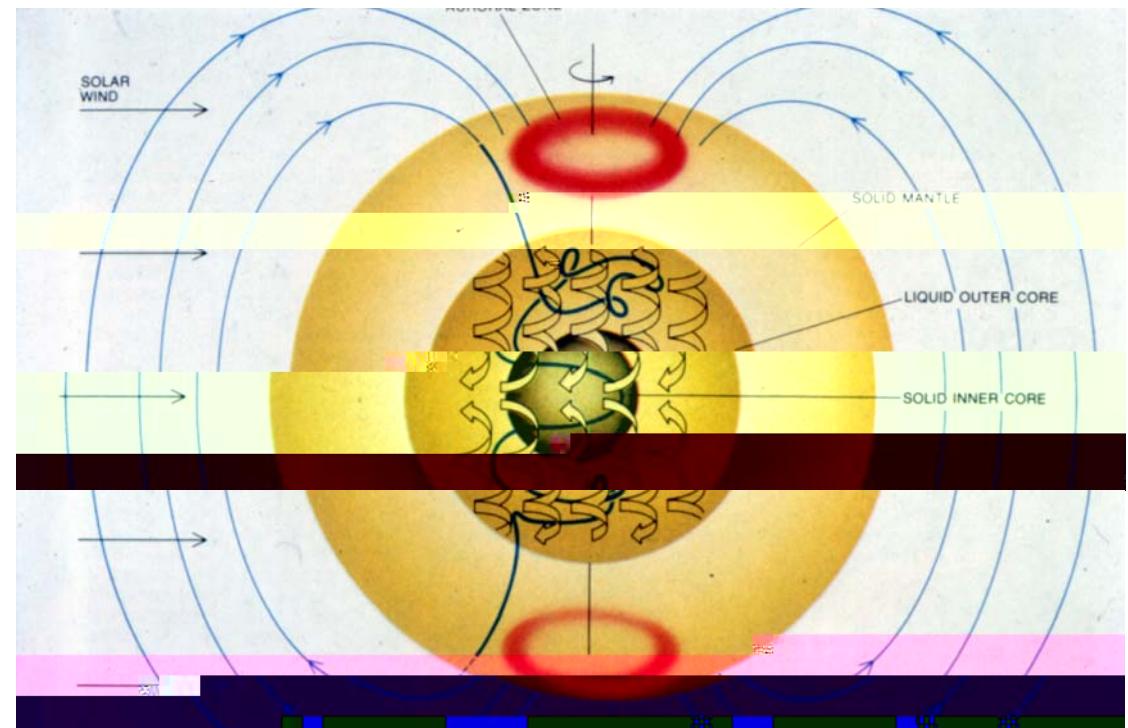
# Phase diagram of ice



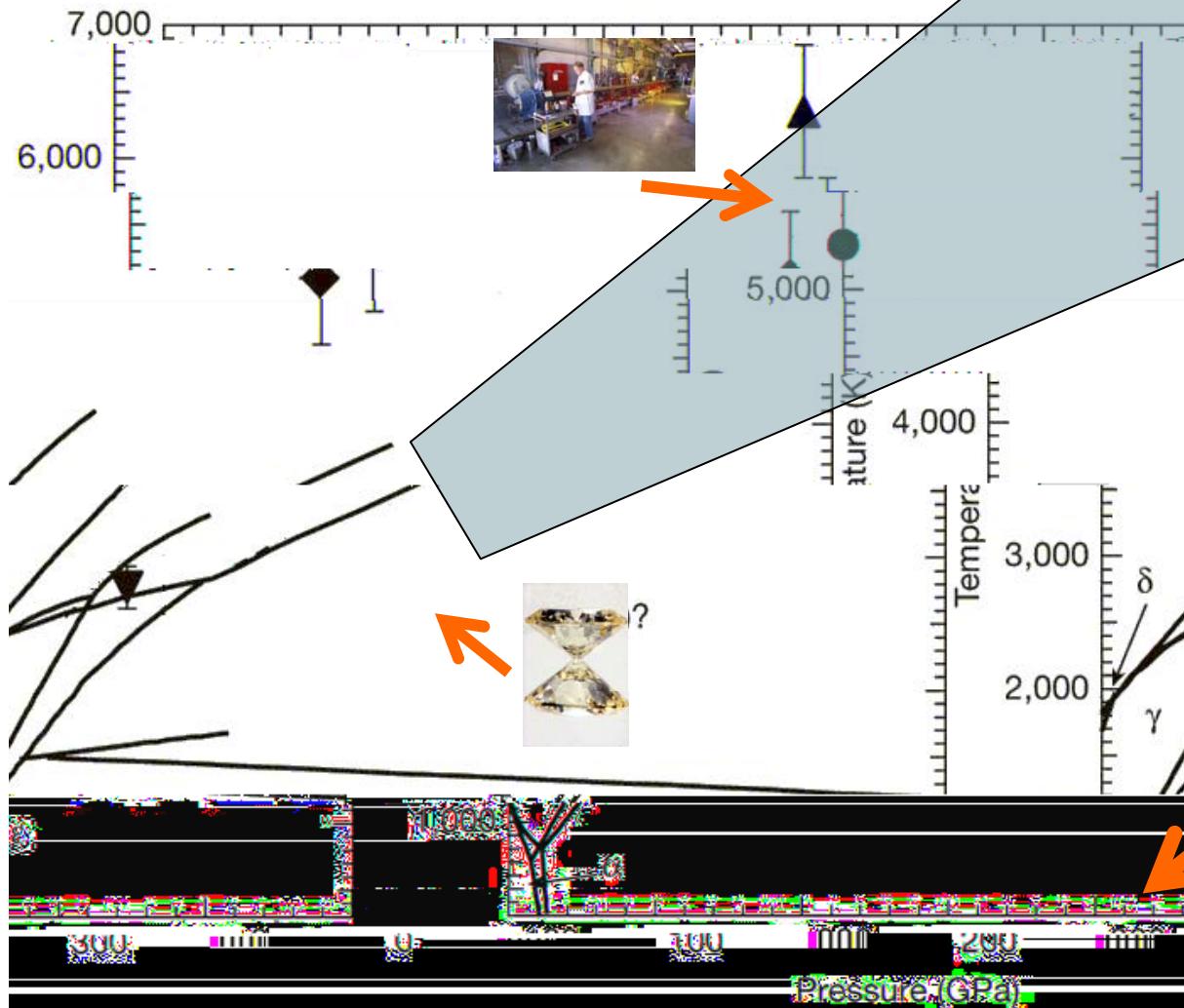


# The importance of the Earth's core

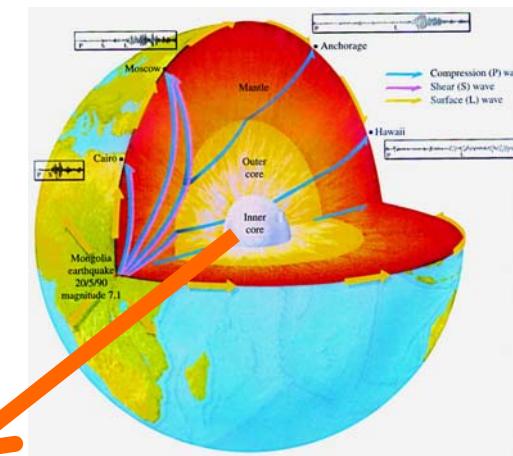
- Contains ~ 30% of the mass and ~ 15% of the volume of the Earth
- The Earth's core is the seat of major global processes.
- Convection in the outer core generates the Earth's magnetic field.
- Heat flow from the core helps drive Mantle convection.



# Phase diagram of Fe



Picture adapted from:  
Nguyen and Holmes,  
Nature, **427**, 339 (2004)



ICB  $P = 330$  GPa

# Theory

- Statistical mechanics
  - Free energies
  - Coexistence of phases
  - Coexistence of phases + free energies
- Interatomic interactions
  - Empirical potentials
  - Density functional theory
  - Quantum Monte Carlo

# Quantum mechanics (Schrödinger equation, 1926)



$$\begin{aligned}H\psi &= E\psi \\H &= T + V \\\psi &= \psi(r_1, \dots, r_N)\end{aligned}$$





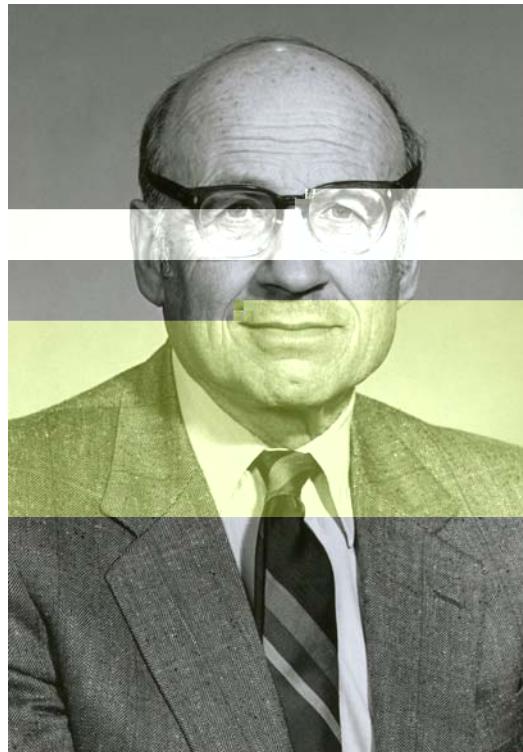
*“The fundamental laws governing most of physics and all of chemistry are now completely known. The only problem is that the solution is much too difficult to be practicable.”*

Paul Dirac  
1929

# Density Functional theory

Hohenberg & Kohn 1964

Kohn & Sham 1965



$$H\psi = E\psi$$

$$\psi(r_1, \dots, r_N) \downarrow n(r)$$

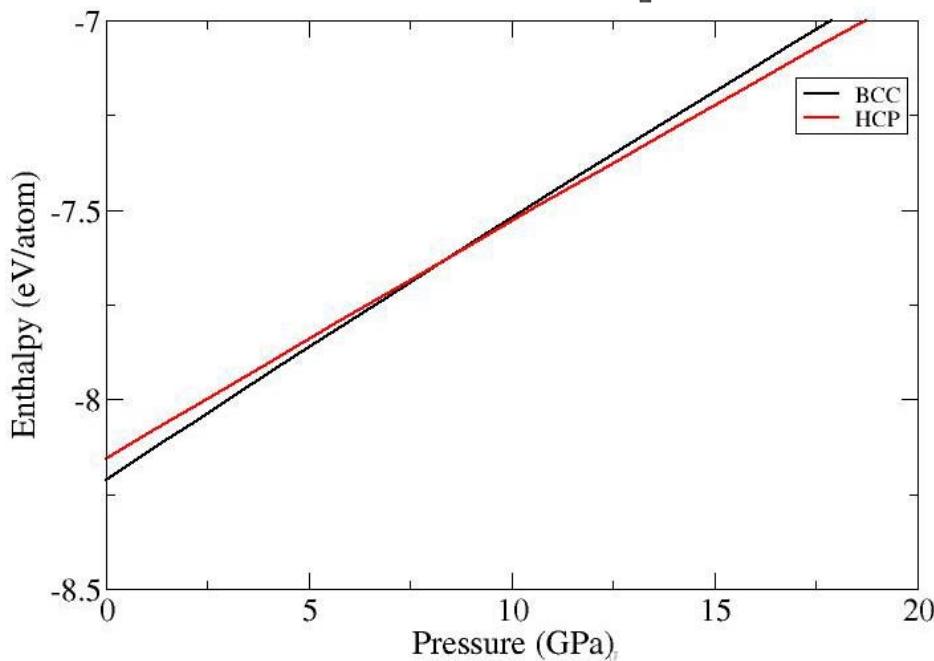
$$H_{KS} \psi_i = E_i \psi_i \quad i = 1, N$$

$$H_{KS} = T + V + V_H + V_{XC}$$

# Phase stability:

## Zero temperature

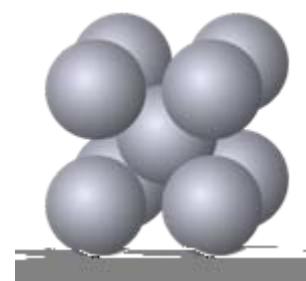
- Energy:
  - Internal energy:  $E(V)$
  - Enthalpy:  $H(p) = E(V) + pV$   
 $p = -dE/dV$



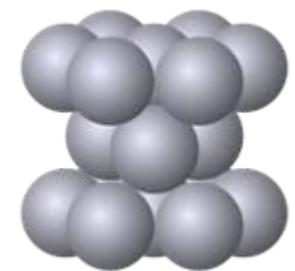
Equation of state (e.g. Birch-Murnaghan, Phys. Rev. **71**, 809 [1947]):

$$E(V) = E_0 + \frac{3}{2} V_0 K_0 \left( \frac{3}{4}(1+2\xi) \left( \frac{V_0}{V} \right)^{4/3} - \frac{\xi}{2} \left( \frac{V_0}{V} \right)^2 - \frac{3}{2}(1+\xi) \left( \frac{V_0}{V} \right)^{2/3} + \frac{1}{2} \xi + \frac{3}{2} \right)$$

Body centred cubic (bcc)



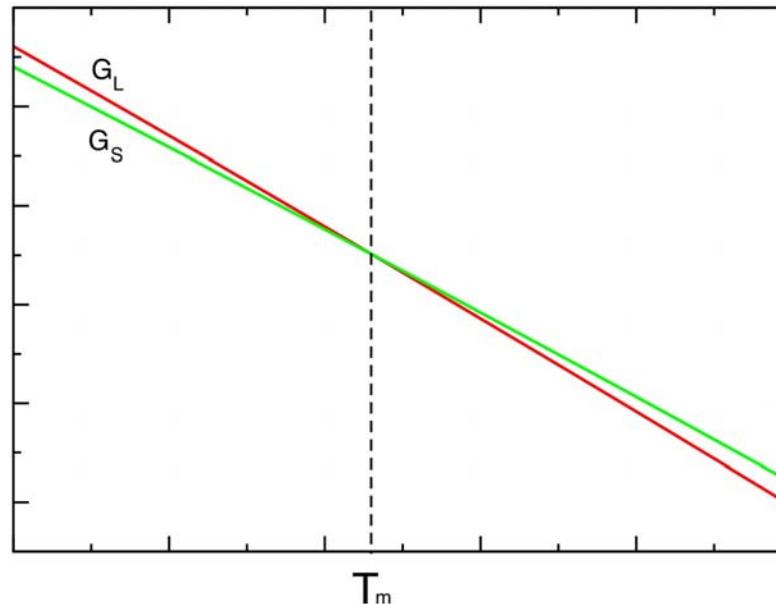
Hexagonal closed packed (hcp)



# Finite temperature

- Free Energy:
  - Helmholtz free energy:  $F(V, T) = E(V, T) - TS(V, T)$
  - Gibbs free energy:  $G(p, T) = F(V, T) + pV$   
 $p = -dF/dT$

e.g. Melting:



# The Helmholtz free energy

Solids: Low T

$$F(V, T) = F_{perf}(V, T) + F_{harm}(V, T) + \cancel{F_{anharm}(V, T)}$$

$$F_{harm}(V, T) = 3k_B T \frac{1}{N_{\mathbf{q},s}} \sum_{\mathbf{q},s} \ln 2 \sinh \frac{\omega_{\mathbf{q},s}(V, T)}{2k_B T}$$

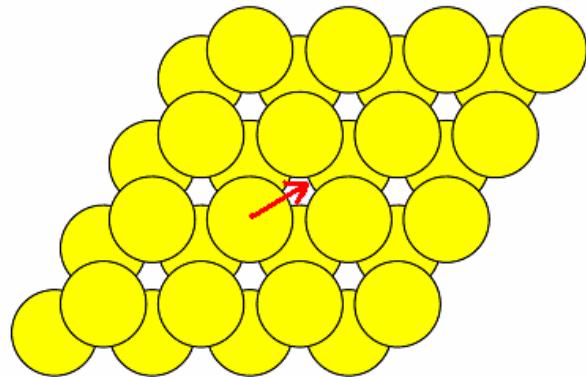
Dynamical matrix:  $D(\mathbf{q}) = \frac{1}{M} \sum_{\mathbf{R}} \Phi(\mathbf{R}) e^{i\mathbf{q}\cdot\mathbf{R}}$

Force constant matrix:  $F_\alpha(\mathbf{u}) = \sum_{\mathbf{q}} \alpha(\mathbf{q}) u(\mathbf{q}) u(\mathbf{q})^\dagger$

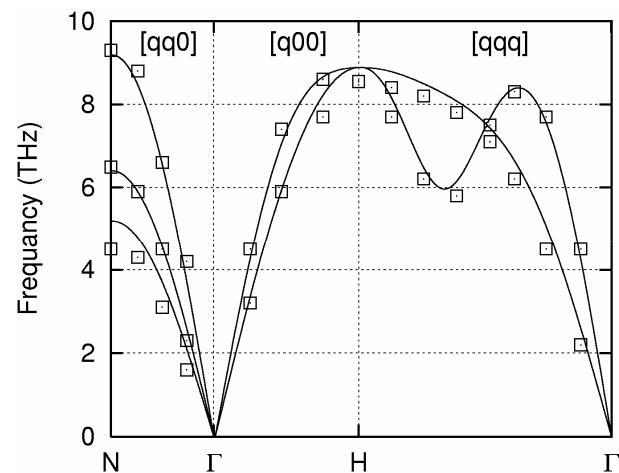
# Small displacement method:

PHON code, freely available at:  
<http://chianti.geol.ucl.ac.uk/~dario>

D. Alfè Comp. Phys. Comm. 180 2622 (2009)



## DFT phonons of Fe:



# The Helmholtz free energy

Solids: High T

$$F(V,T) = F_{perf}(V,T) + F_{harm}(V,T) + \cancel{F_{anharm}(V,T)}$$

$$F_{harm}(V,T) = 3k_B T \frac{1}{N_{\mathbf{q},s}} \sum_{\mathbf{q},s} \ln 2 \sinh \frac{\omega_{\mathbf{q},s}(V,T)}{2k_B T}$$

Liquids:

$$F(V,T) = -k_B T \ln \frac{1}{N! \Lambda^{3N}} \int_V dR e^{-U(R)/k_B T}$$

# Thermodynamic integration

$$U_{ref}, F_{ref}$$

$$U_\lambda = (1 - \lambda)U_{ref} + \lambda U$$

$$F_\lambda = -k_B T \ln \frac{1}{N! \Lambda^{3N}} \int_V dR e^{-U_\lambda(R)/k_B T}$$

$$F - F_{ref} = \int_0^1 d\lambda \frac{dF_\lambda}{d\lambda}$$

$$\frac{dF_\lambda}{d\lambda} = \frac{\int_V dR \frac{\partial U_\lambda}{\partial \lambda} e^{-U_\lambda(R)/k_B T}}{\int_V dR e^{-U_\lambda(R)/k_B T}} = \left\langle \frac{\partial U_\lambda}{\partial \lambda} \right\rangle_\lambda = \left\langle U - U_{ref} \right\rangle_\lambda$$

$$F = F_{ref} + \int_0^1 d\lambda \left\langle U - U_{ref} \right\rangle_\lambda$$

# Thermodynamic integration

$$F = F_{ref} + \int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda$$

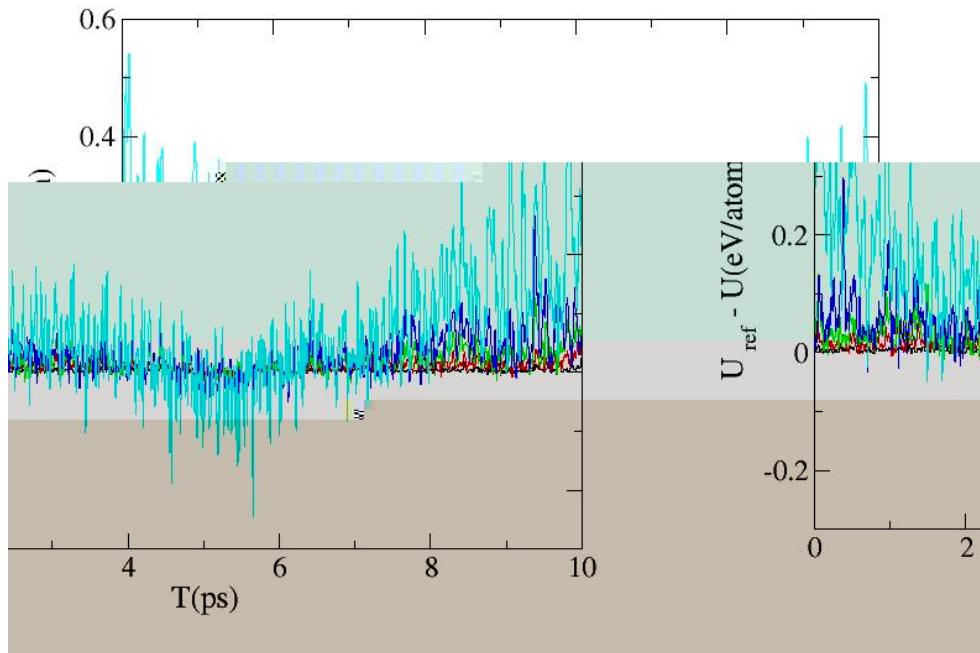
$$F = F_{ref} + \int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda = F_{ref} +$$

# Example: anharmonic free energy of solid Fe at ~350 GPa

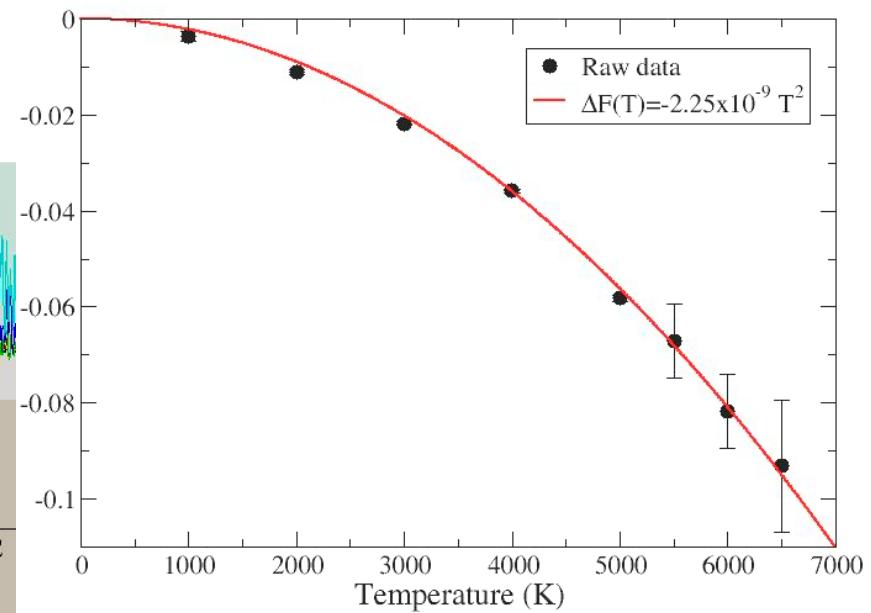
$$F = F_{harm} + \int_0^T dt \frac{d\lambda}{dt} (U - U_{harm})_\lambda$$

$$U_{harm} = \frac{1}{2} \sum_{i\alpha, j\beta} u_{i\alpha} \Phi_{i\alpha, j\beta} u_{j\beta}$$

T=1000, 2000, 3000, 4000, 6500 K



Anharmonic free energy of Fe at V=6.97 Å<sup>3</sup>/atom



# Improving the efficiency of TI

$$F = F_{ref} + \int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda$$

$F$  is independent on the choice of  $U_{ref}$ , but for efficiency choose  $U_{ref}$  such that:

$$\langle (U - U_{ref} - \langle U - U_{ref} \rangle)^2 \rangle$$

is minimum. For solid iron at Earth's core conditions a good  $U_{ref}$  is:

$$U_{ref} = c_1 U_{harm} + c_2 U_{IP}$$

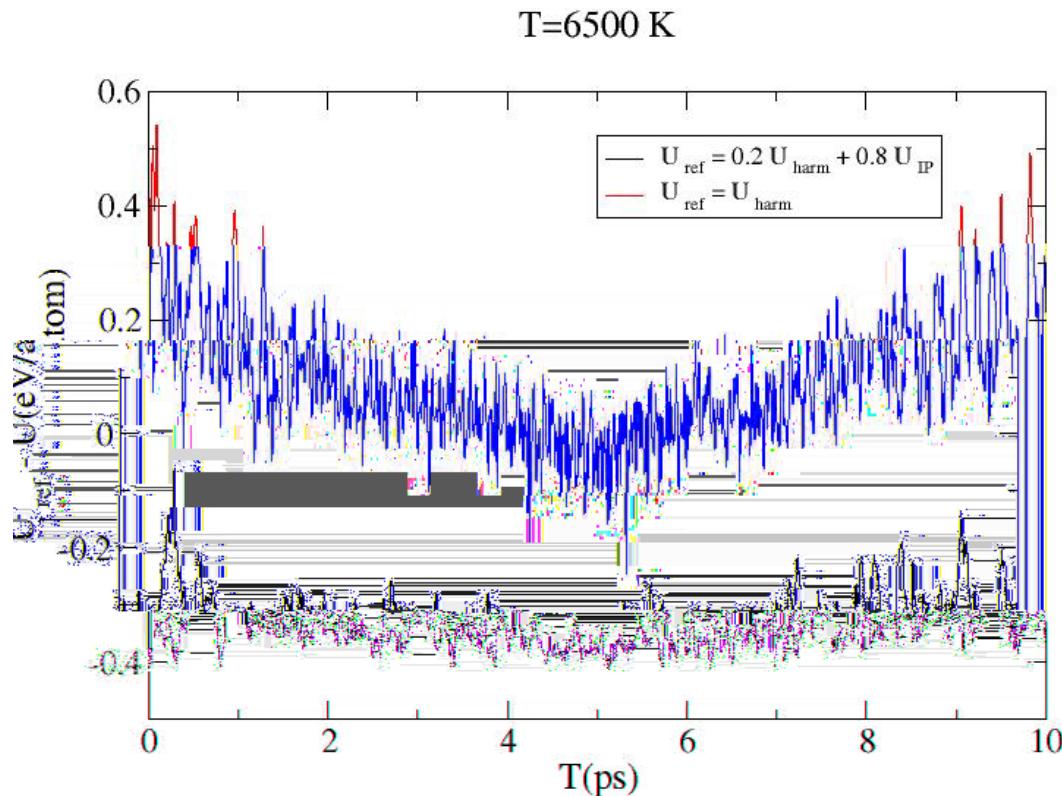
$$U_{harm} = \frac{1}{2} \sum_{i\alpha, j\beta} u_{i\alpha} \Phi_{i\alpha, j\beta} u_{j\beta}$$

$$U_{IP} = \frac{1}{2} \sum_{i \neq j} \frac{A}{|r_i - r_j|^B}; \quad B = 5.86$$

# Improving the efficiency of TI (2)

$$U_{ref} = c_1 U_{harm} + c_2 U_{IP}$$

At high temperature we find  $c_1 = 0.2$ ,  $c_2 = 0.8$



# Thermodynamic integration, a perturbative approach:

$$F = F_{ref} + \int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda$$

$$\langle U - U_{ref} \rangle_\lambda = \left. \langle U - U_{ref} \rangle_{\lambda=0} + \lambda \frac{\partial \langle U - U_{ref} \rangle_\lambda}{\partial \lambda} \right|_{\lambda=0} + o(\lambda^2)$$

$$\frac{\partial \langle U - U_{ref} \rangle_\lambda}{\partial \lambda} = \frac{\partial}{\partial \lambda} \frac{\int_V dR \frac{\partial U_\lambda}{\partial \lambda} e^{-U_\lambda(R)/k_B T}}{\int_V dR e^{-U_\lambda(R)/k_B T}} =$$

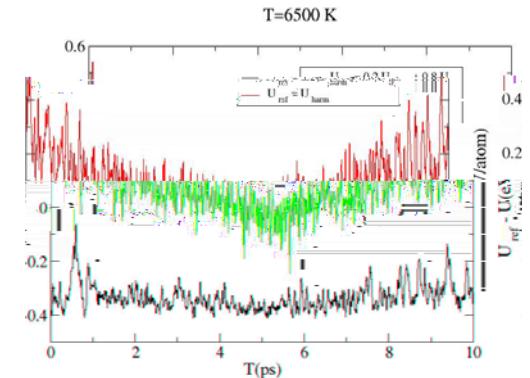
$$-\frac{1}{k_B T} \frac{\int_V dR \frac{\partial U_\lambda}{\partial \lambda}^2 e^{-U_\lambda(R)/k_B T}}{\int_V dR e^{-U_\lambda(R)/k_B T}} - \frac{\int_V dR \frac{\partial U_\lambda}{\partial \lambda} e^{-U_\lambda(R)/k_B T}}{\int_V dR e^{-U_\lambda(R)/k_B T}}^2 = -\frac{1}{k_B T} \langle \delta \Delta U_\lambda^2 \rangle_\lambda$$

$$\delta\Delta U_\lambda = U - U_{ref} - \langle U - U_{ref} \rangle_\lambda$$

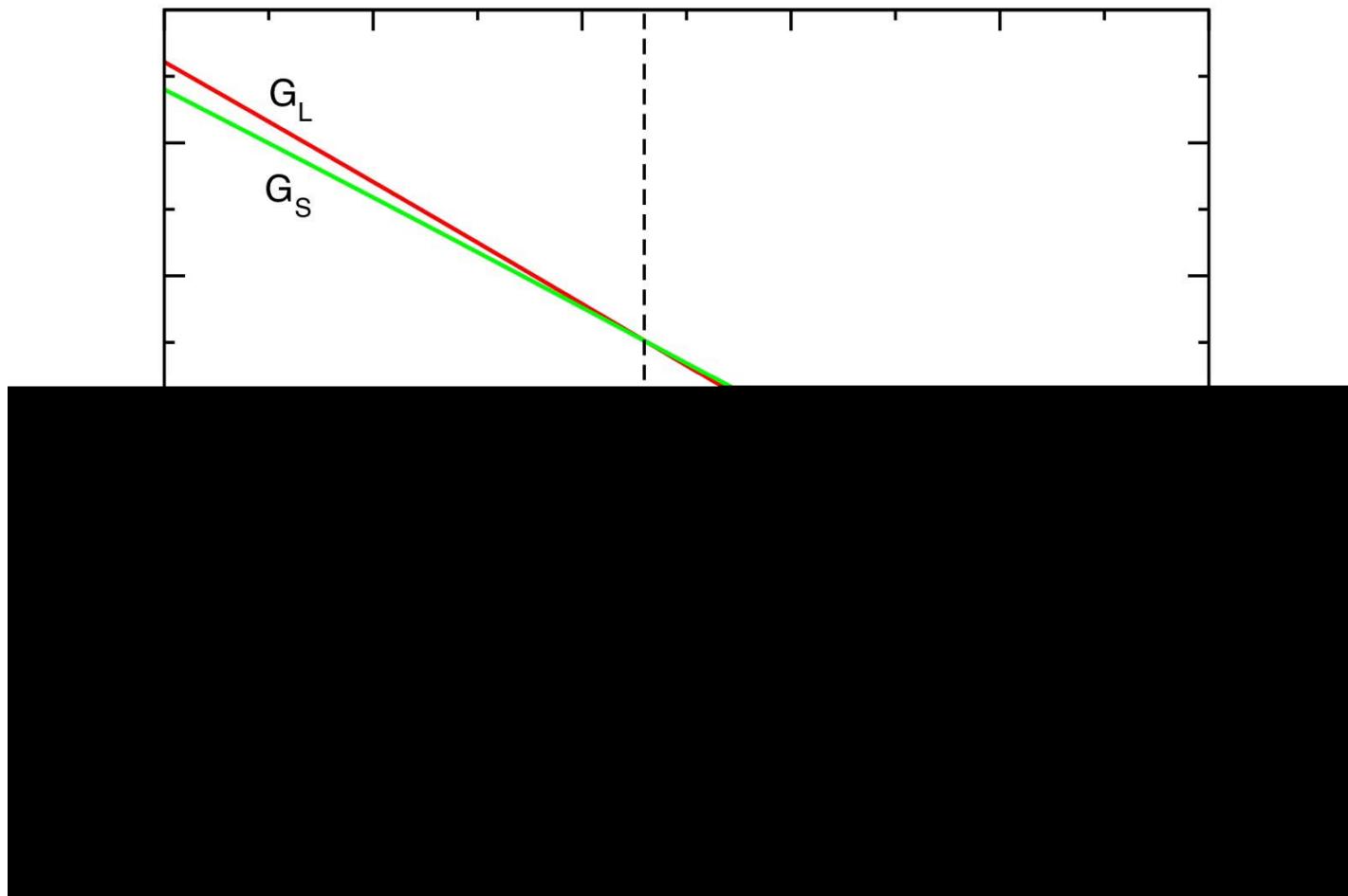
$$\langle U - U_{ref} \rangle_\lambda = \langle U - U_{ref} \rangle_{\lambda=0} - \frac{\lambda}{k_B T} \langle \delta\Delta U_0^2 \rangle_0 + o(\lambda^2)$$

$$\int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda = \langle U - U_{ref} \rangle_{\lambda=0} - \frac{1}{2k_B T} \langle \delta\Delta U_0^2 \rangle_0$$

Only need to run simulations with one potential (the reference potential for example).



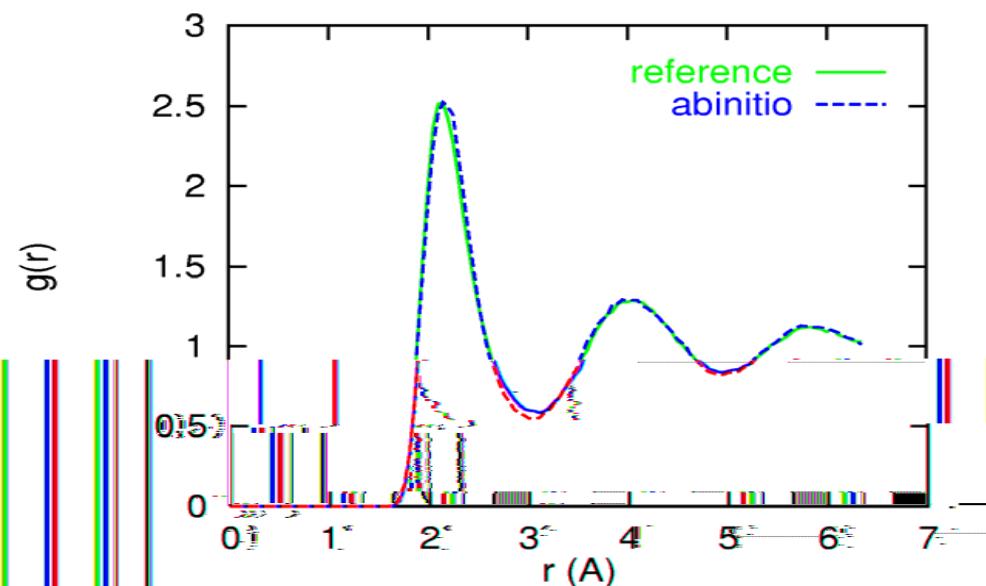
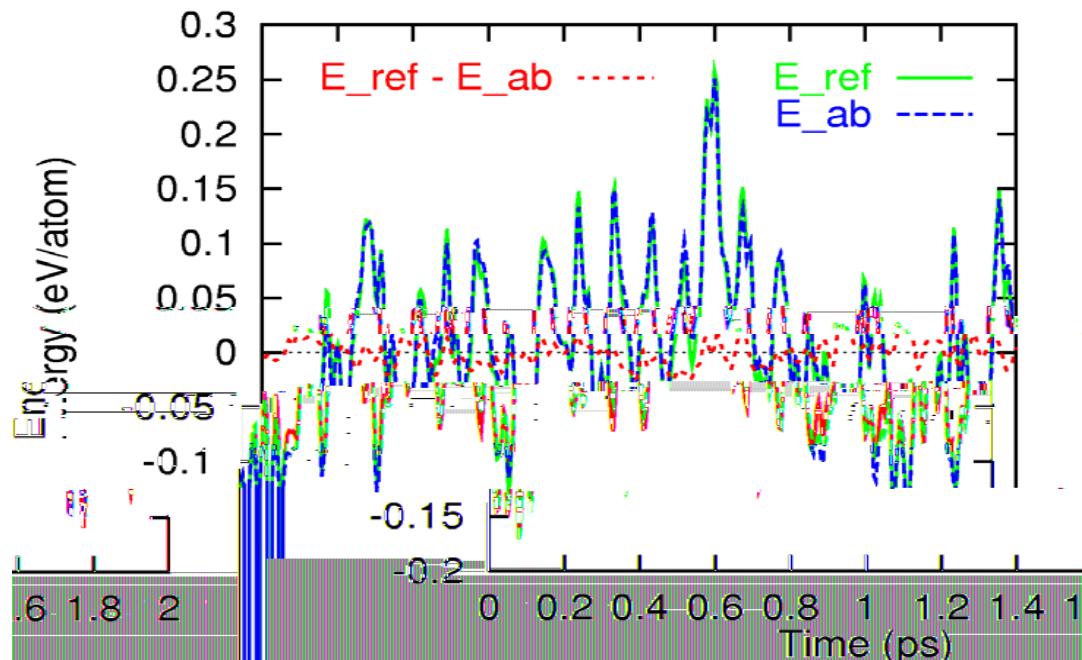
# Melting of Fe



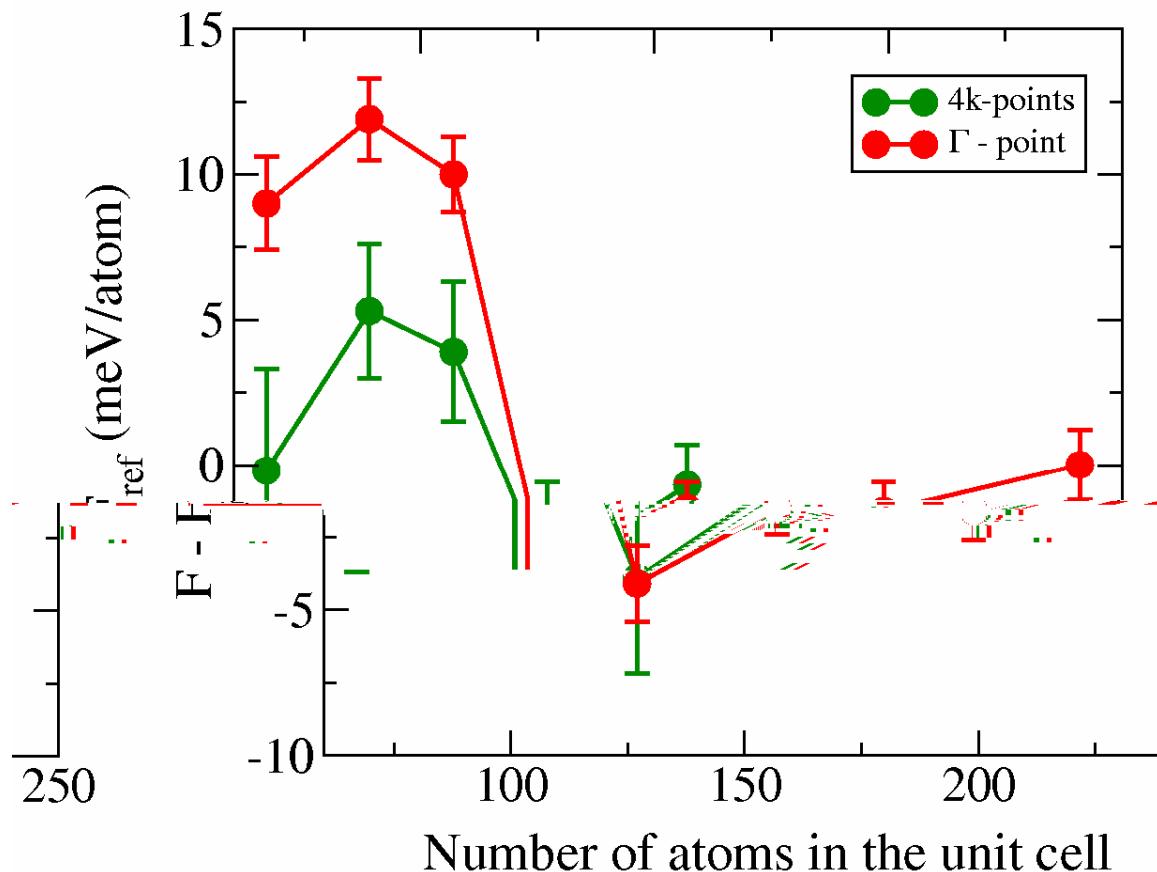
# Liquid Fe

$$U_{ref} = \frac{1}{2} \sum_{i \neq j} \frac{A}{|r_i - r_j|^B}$$

$$B = 5.86$$

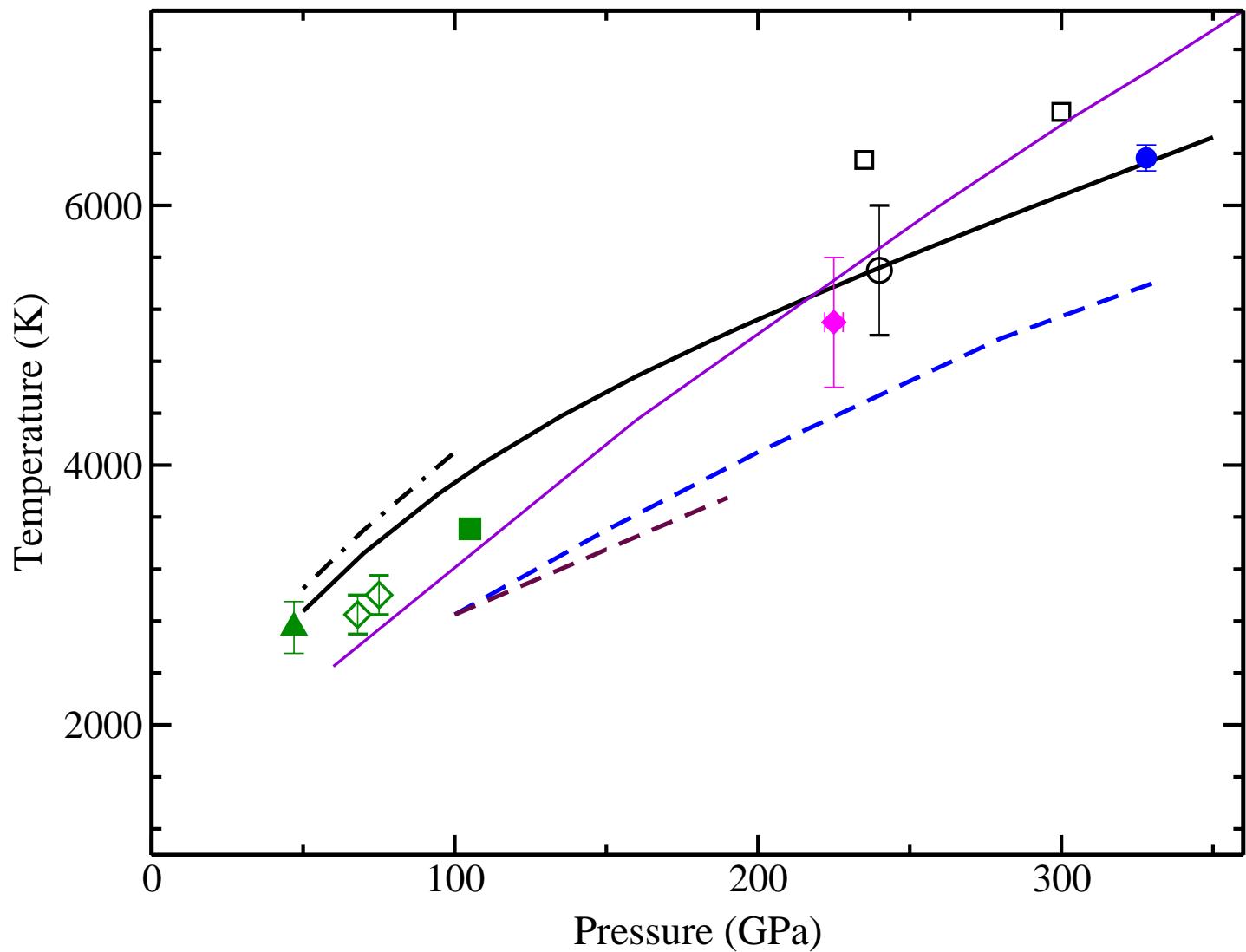


# Size tests

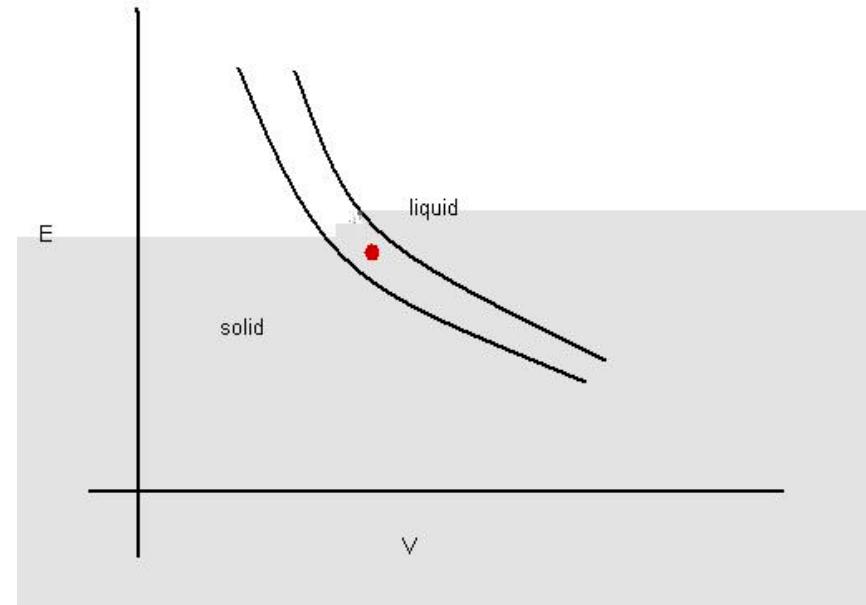
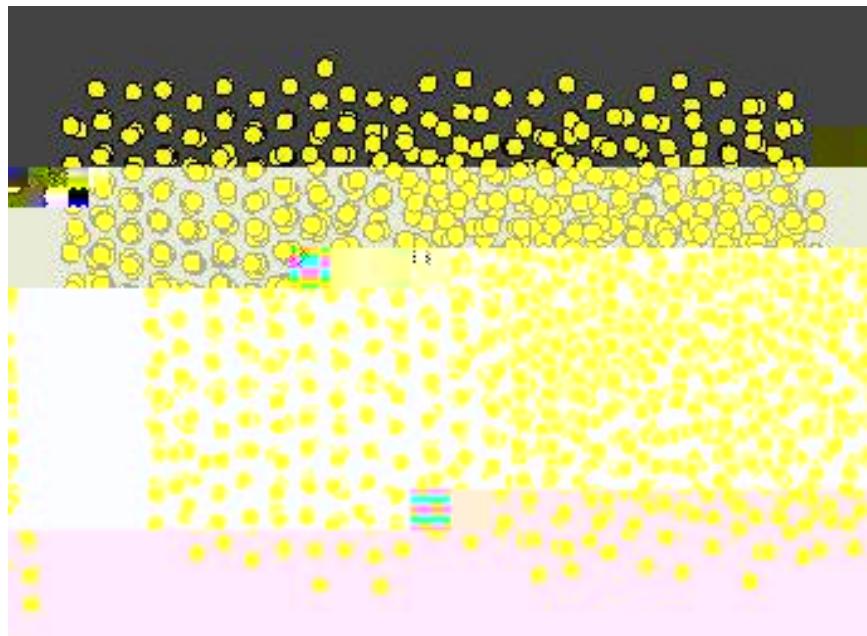


$$\Delta T \approx 100 \text{ K} \rightarrow \Delta G \approx 10 \text{ meV / atom}$$

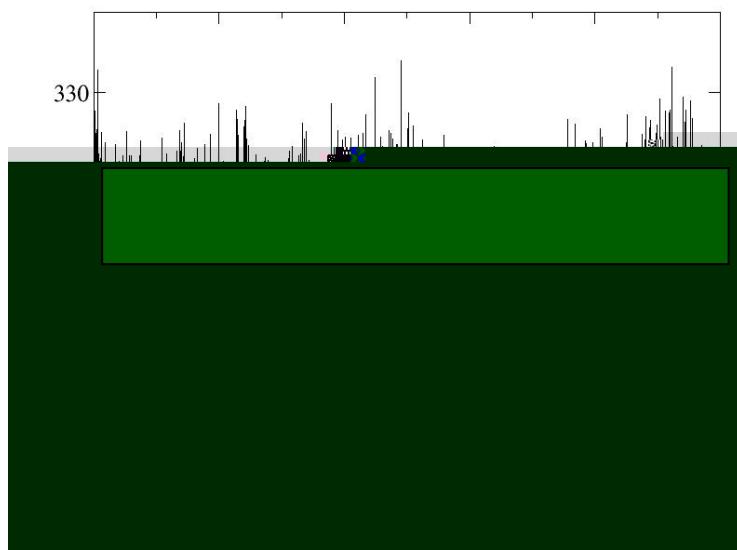
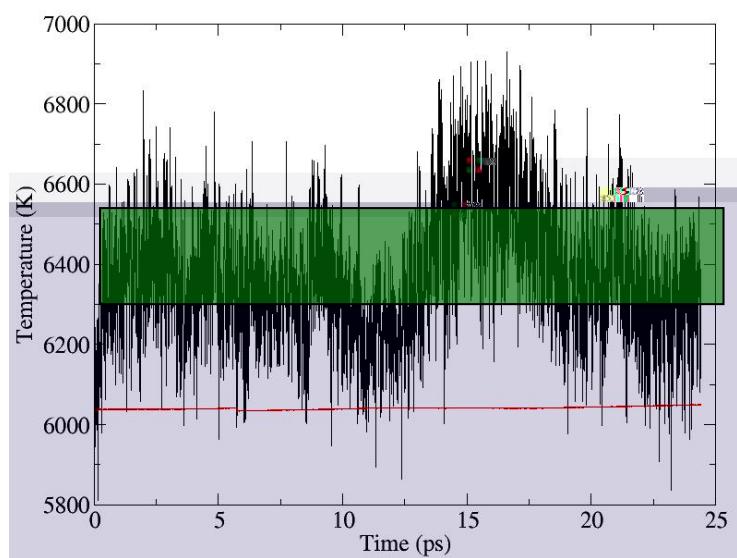
# The melting curve of Fe



# Melting: coexistence of phases

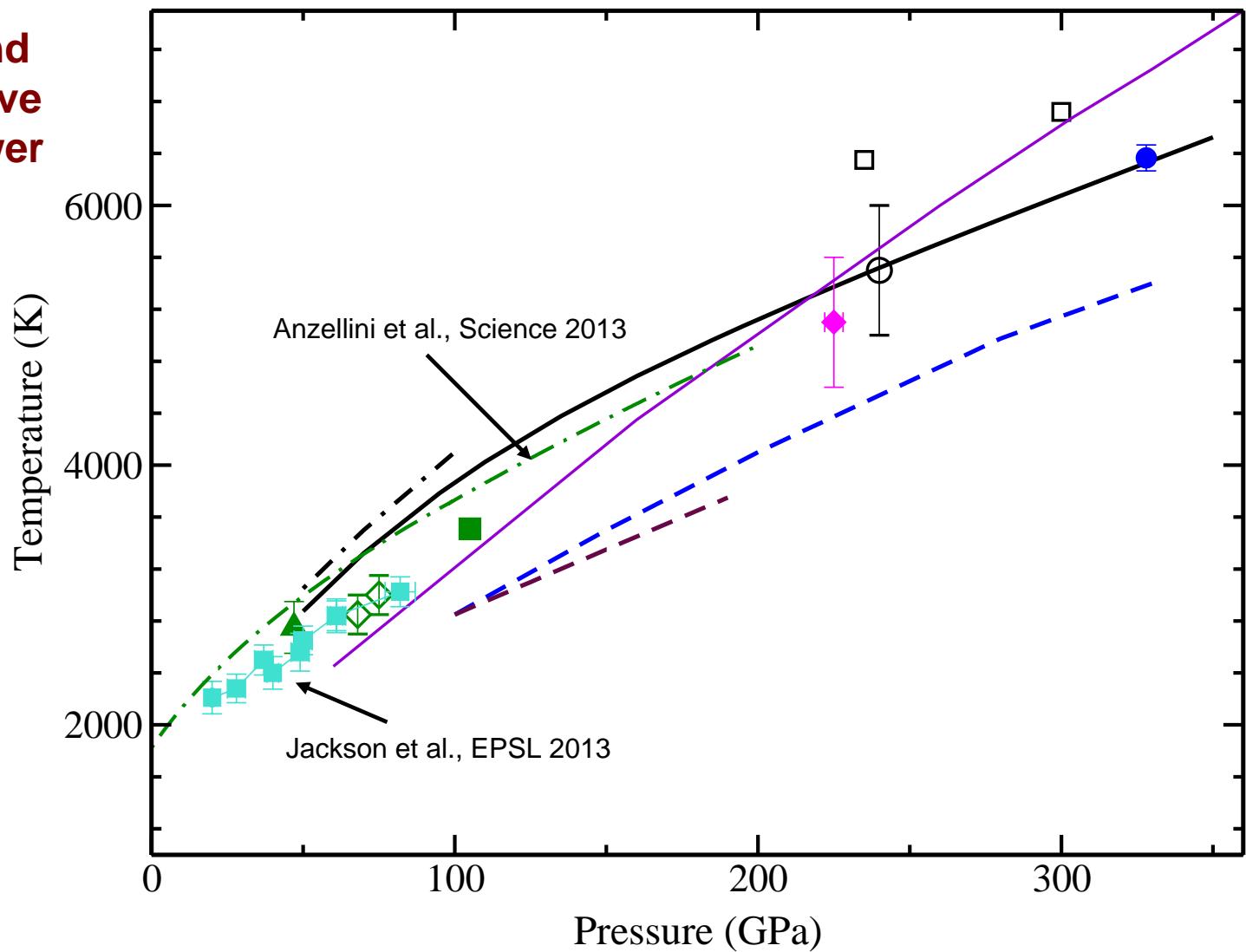


NVE ensemble: for fixed  $V$ , if  $E$  is between solid and liquid values, simulation will give coexisting solid and liquid

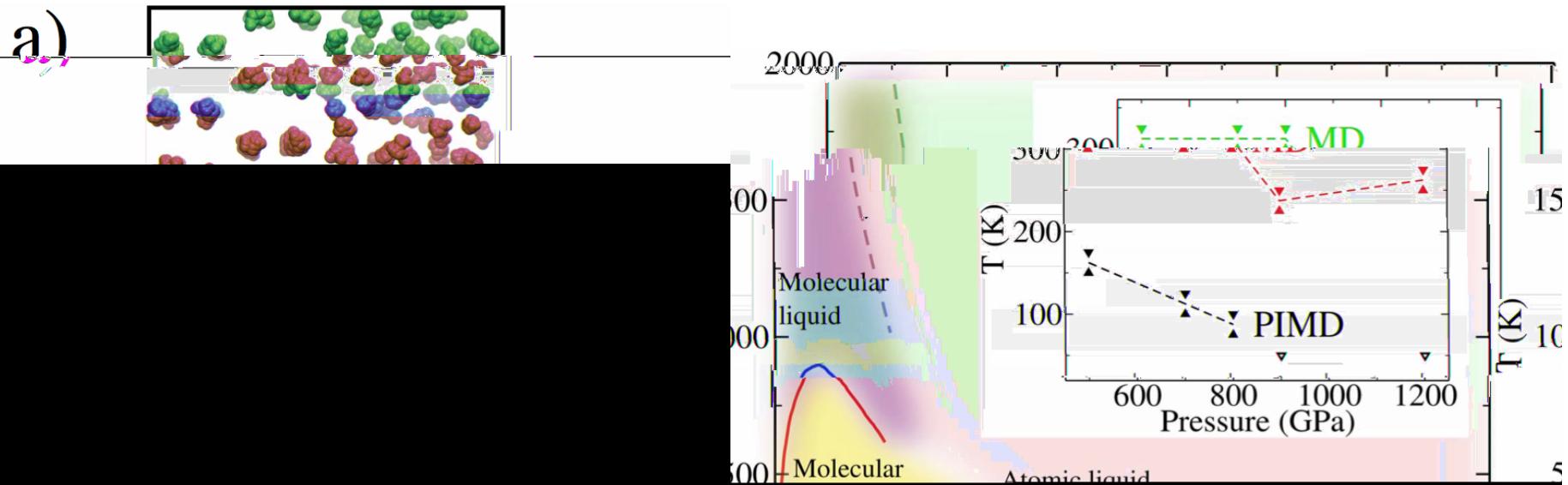


# The melting curve of Fe

Free energy and coexistence give the same answer



# Melting curve of H



# Theory

- Statistical mechanics
  - Free energies
  - Coexistence of phases
  - Coexistence of phases + free energies
- Interatomic interactions
  - Empirical potentials
  - Density functional theory
  - Quantum Monte Carlo

# Quantum Monte Carlo

Variational Monte Carlo:

Energy  $E_V$  depends on  $T$

$$E_V = \frac{\int \Psi_T^*(\mathbf{R}) \hat{H} \Psi_T(\mathbf{R}) d\mathbf{R}}{\int \Psi_T^*(\mathbf{R}) \Psi_T(\mathbf{R}) d\mathbf{R}} \geq E_0$$

Diffusion Monte Carlo:

Extracting the ground state: substitute  $t = it$

$$-\frac{\partial \phi(\mathbf{x}, t)}{i\partial t} = (\hat{H} - E_T) \phi(\mathbf{x}, t)$$

$$-\frac{\partial \phi(\mathbf{x}, \tau)}{\partial \tau} = \frac{1}{2} \sum_{i=1}^N \Delta_i \phi(\mathbf{x}, \tau) + (V - E_T) \phi(\mathbf{x}, \tau)$$

$$\tau \rightarrow \infty, \quad \phi(\mathbf{x}, \tau) \rightarrow \Phi_0(\mathbf{x})$$

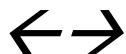
Imaginary time Schroedinger equation with  $V = 0$ : Diffusion equation

$$-\frac{\partial \phi(\mathbf{x}, \tau)}{\partial \tau} = \frac{1}{2} \sum_{i=1}^N \Delta_i \phi(\mathbf{x}, \tau)$$

$\downarrow$  Rate equation

$$-\frac{\partial \phi(\mathbf{x}, \tau)}{\partial \tau} = (V - E_T) \phi(\mathbf{x}, \tau)$$

*Diffusing particles (walkers) with birth/death process*



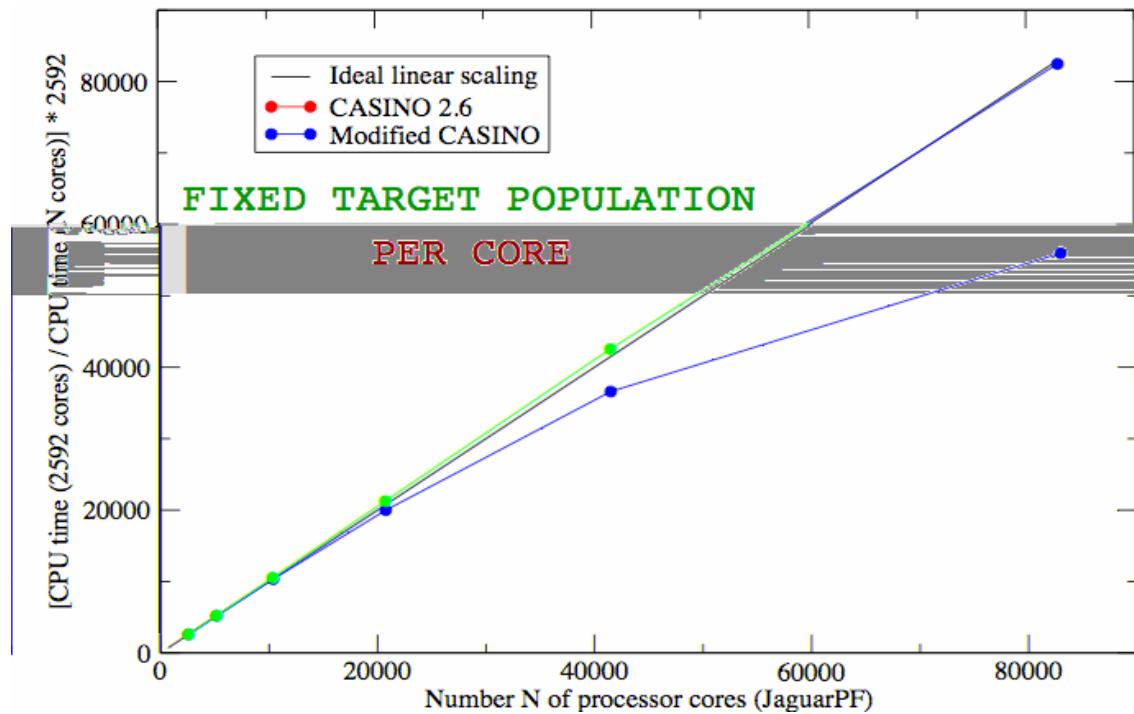
*distribution function*

Approximations:

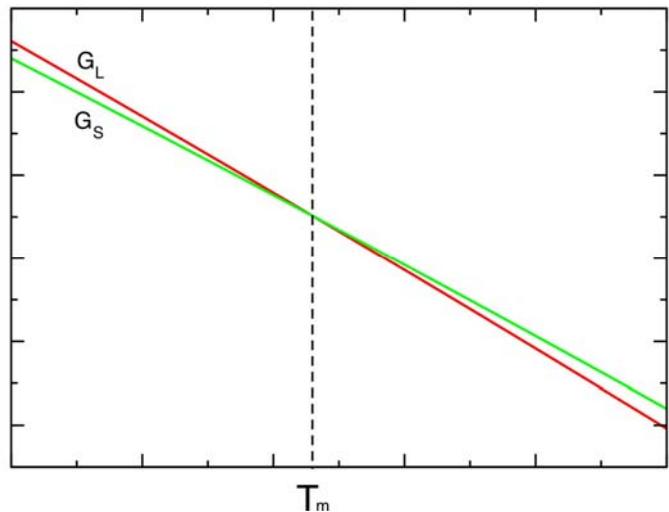
- Fixed nodes approximation:
- Pseudopotentials (locality approximation)

DMC is  $\sim 10^4$  times more expensive than DFT

# QMC scaling on JaguarPF (Cray XT6, 300,000 cores at ONRL)



# Melting of Fe from QMC:



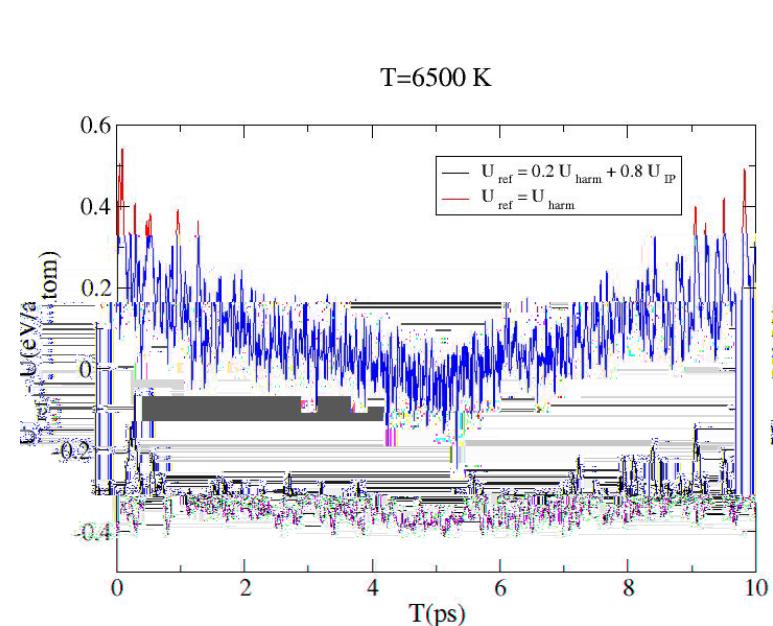
Free energy corrections from DFT to QMC:

$$\delta T_m = \frac{\Delta G^{ls}(T_m^{ref})}{S_{ref}^{ls}}$$

# Thermodynamic integration, a perturbative approach:

$$F = F_{ref} + \int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda$$

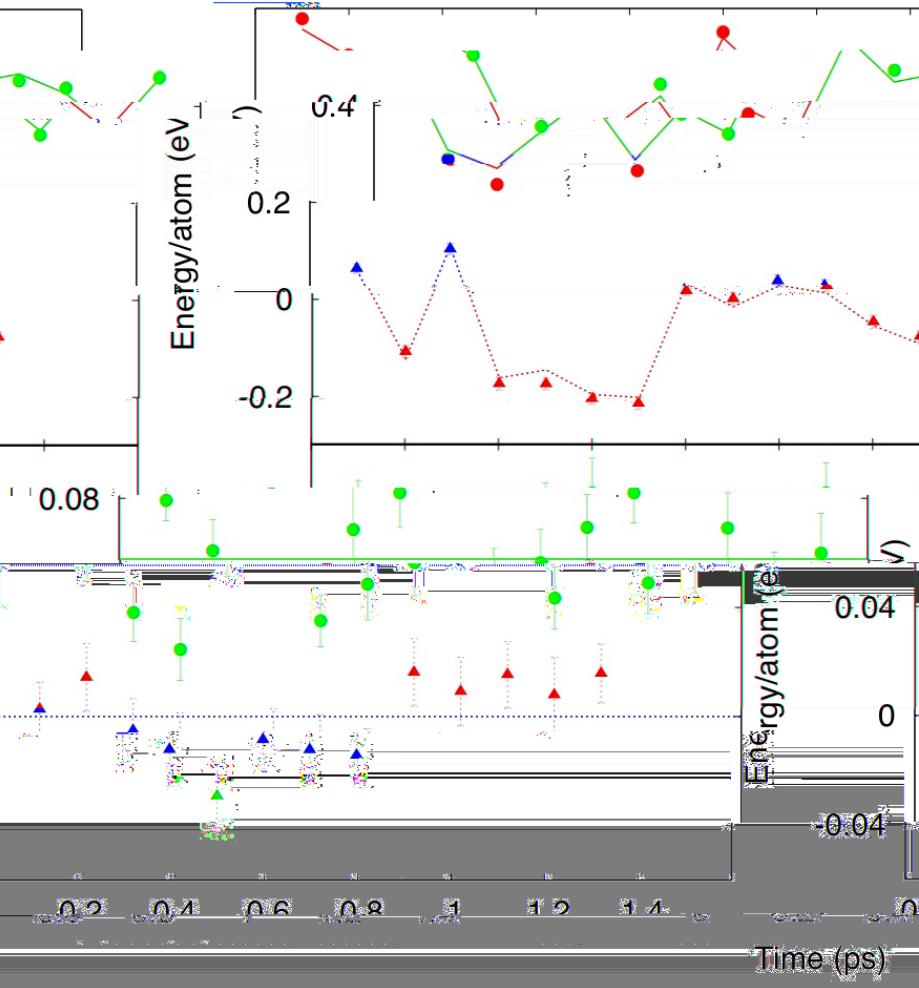
$$\langle U - U_{ref} \rangle_\lambda = \left. \langle U - U_{ref} \rangle_{\lambda=0} + \lambda \frac{\partial \langle U - U_{ref} \rangle_\lambda}{\partial \lambda} \right|_{\lambda=0} + o(\lambda^2)$$



$$\int_0^1 d\lambda \langle U - U_{ref} \rangle_\lambda = \langle U - U_{ref} \rangle_{\lambda=0} - \frac{1}{2k_B T} \langle \delta \Delta U_0^2 \rangle_{\lambda=0}$$

$$\delta \Delta U_\lambda = U - U_{ref} - \langle U - U_{ref} \rangle_\lambda$$

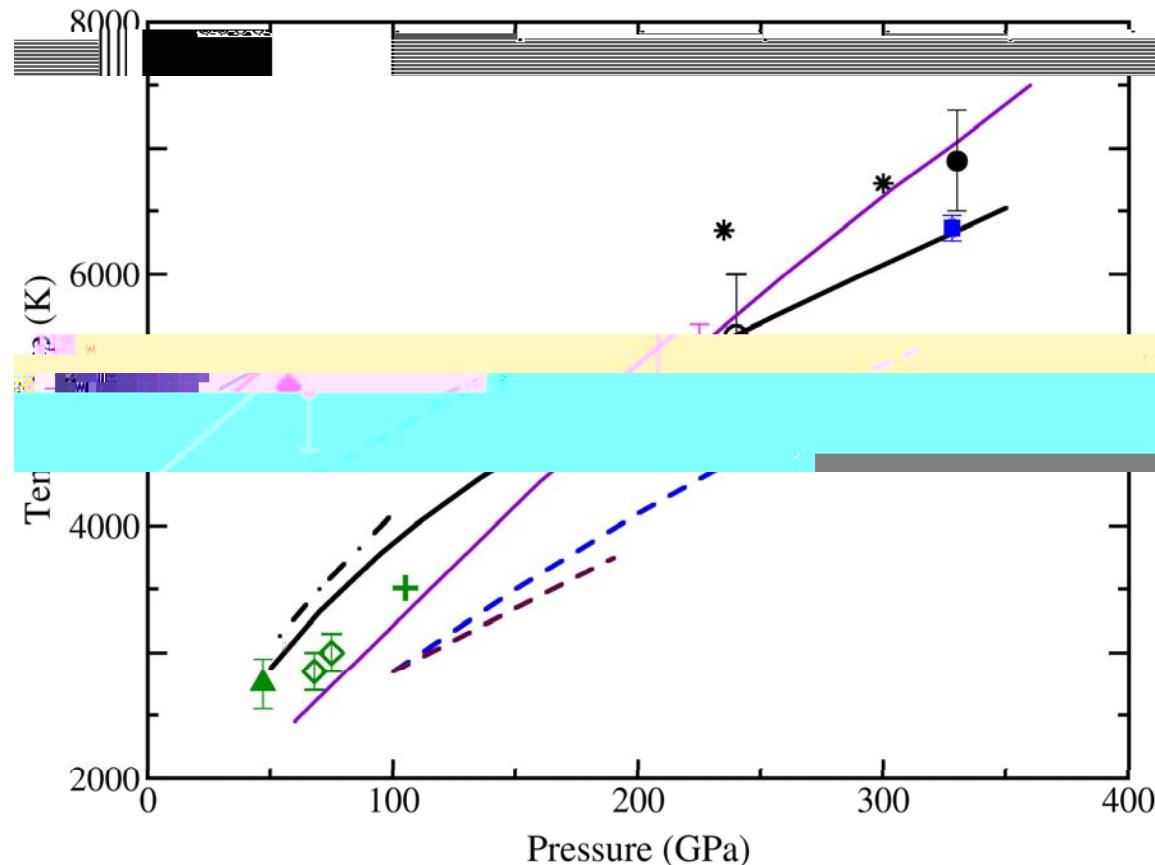
# QMC correction to the DFT Fe melting curve



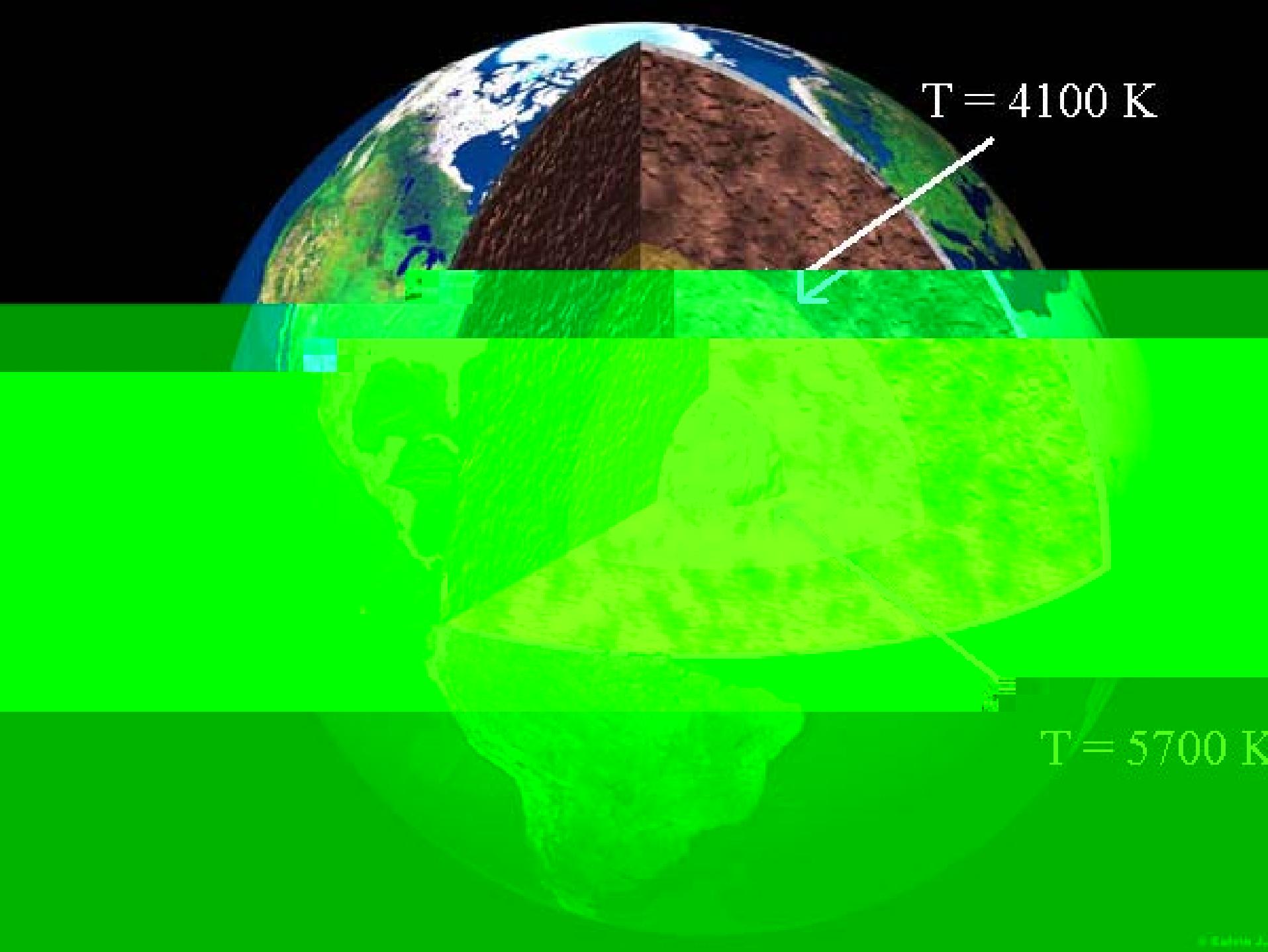
$$\Delta G^{ls}(T_m^{ref}) = 0.05 \pm 0.02 \text{ eV/atom}$$

$$\delta T_m = \frac{\Delta G^{ls}(T_m^{ref})}{S_{ref}^{ls}} = 550 \pm 250$$

# Melting curve of Fe



E. Sola and D. Alfè, Phys. Rev. Lett, **103**, 078501 (2009)



$T = 4100 \text{ K}$

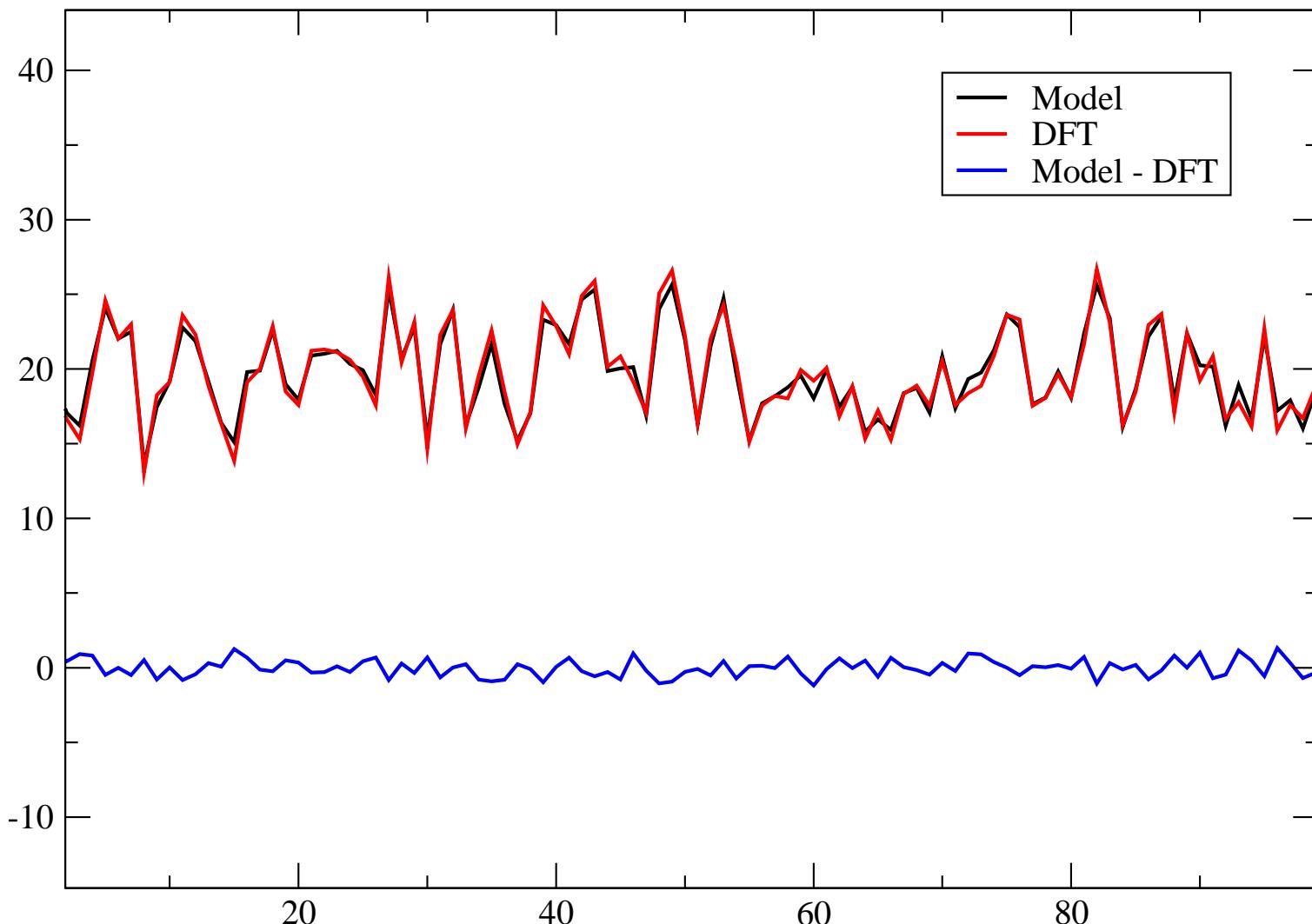
$T = 5700 \text{ K}$

# Theory

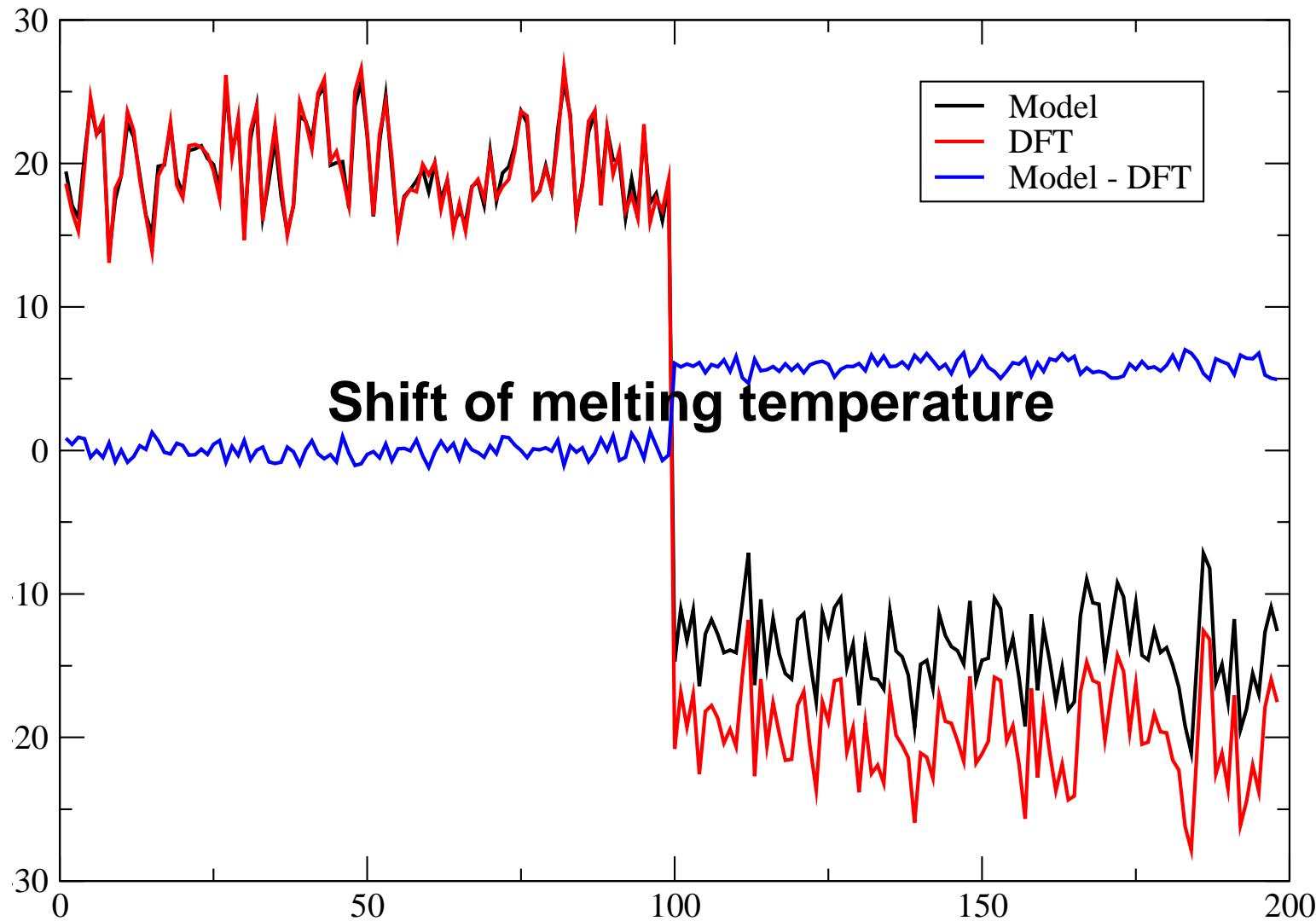
- Statistical mechanics
  - Free energies
  - Coexistence of phases
  - Coexistence of phases + free energies
- Interatomic interactions
  - Empirical potentials
  - Density functional theory
  - Quantum Monte Carlo

# Combining coexistence and free energies

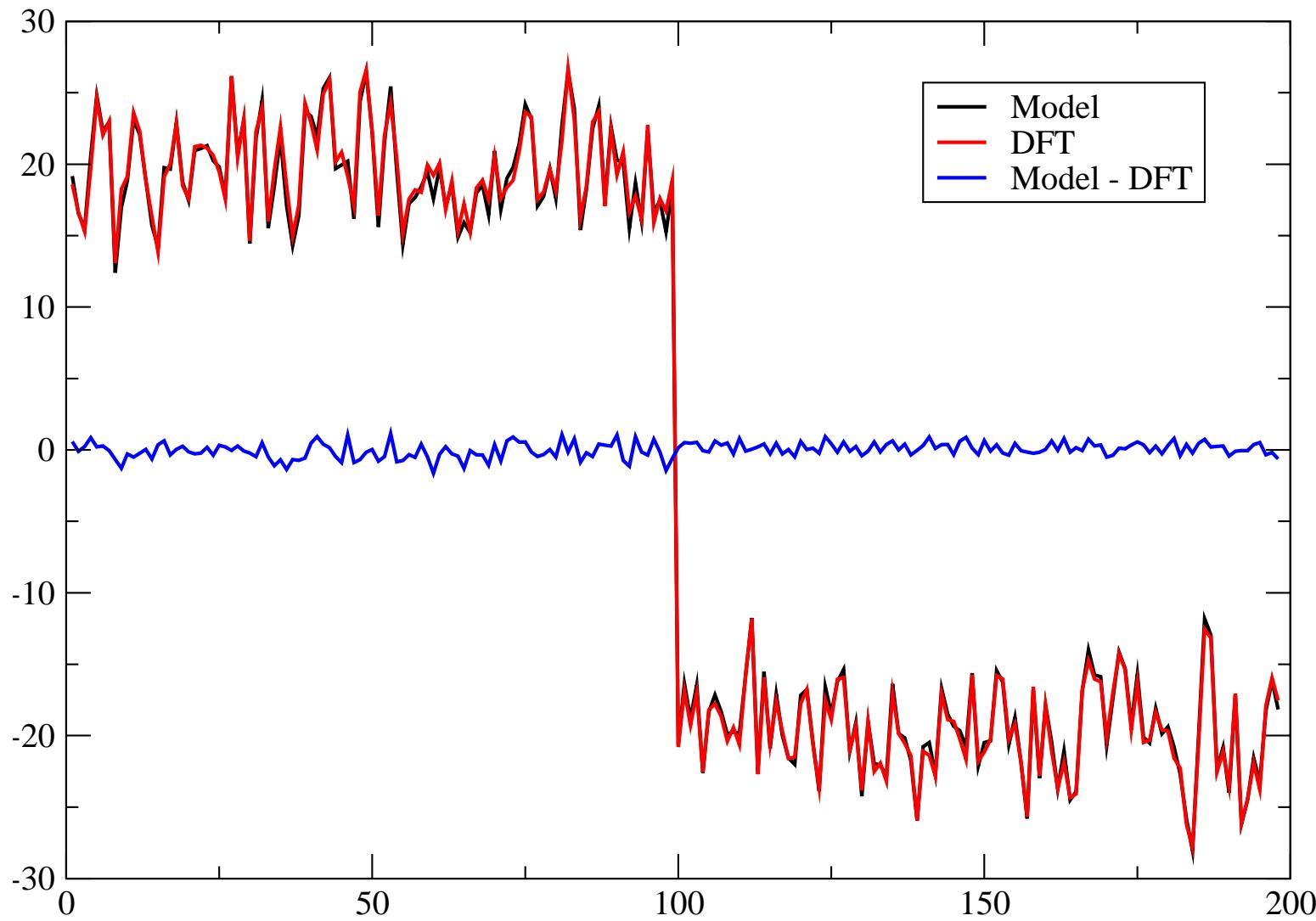
# Fitting a model (e.g. using only the liquid):



# Performance of the model also on the solid:



# Fitting a model (using both liquid and solid):



## Potential fitting summary:

- For best results use data from both phases (liquid and solid, or solid and liquid, or even several different solids if interested in complex phase diagrams).
- If potential is only fitted to one phase then transferability is not guaranteed, and it will usually result in a (possibly large) shift of the phase boundary that needs to be corrected for.

# Strategy for melting of Ta, Mo and Ni:

Coexistence of phases with classical potential:

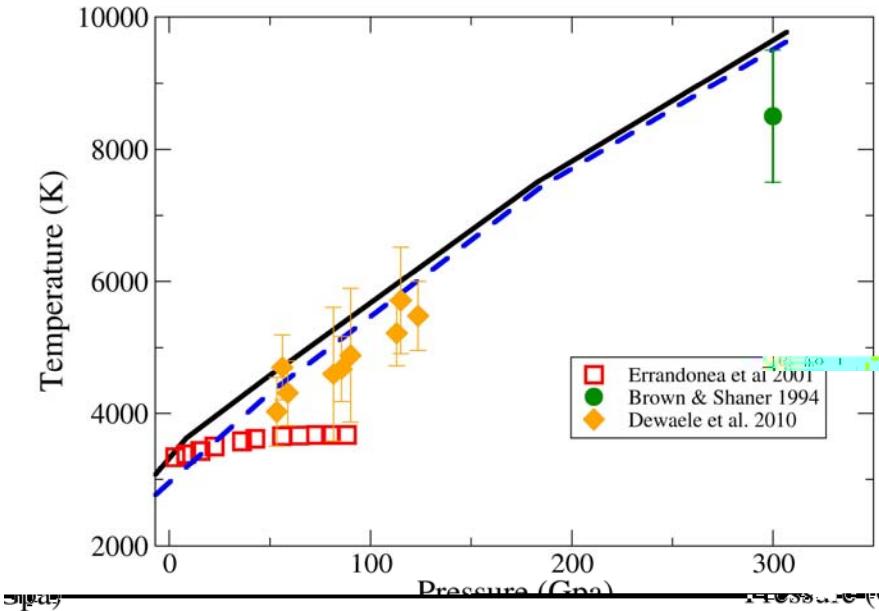
$$U_{\text{ref}}(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N) = \frac{1}{2} \varepsilon \sum_{i \neq j}^n \frac{a}{r_{ij}} - C \varepsilon \sum_i \sum_{j(\neq i)}^m \frac{a}{r_{ij}}^{1/2}$$

Free energy corrections:

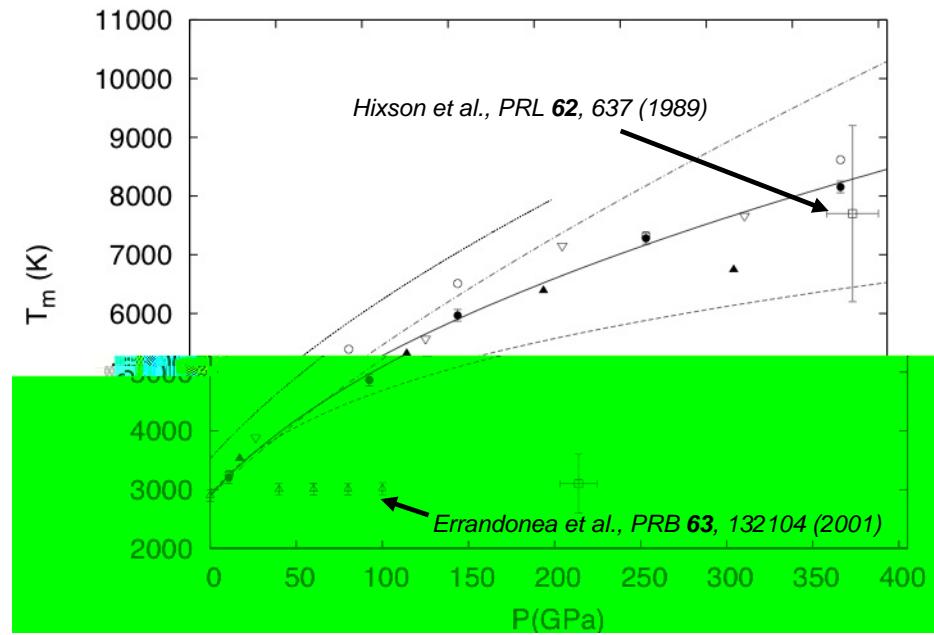
$$\delta T_m = \frac{\Delta G^{ls}(T_m^{ref})}{S_{ref}^{ls}}$$

# Melting curves of Ta and Mo

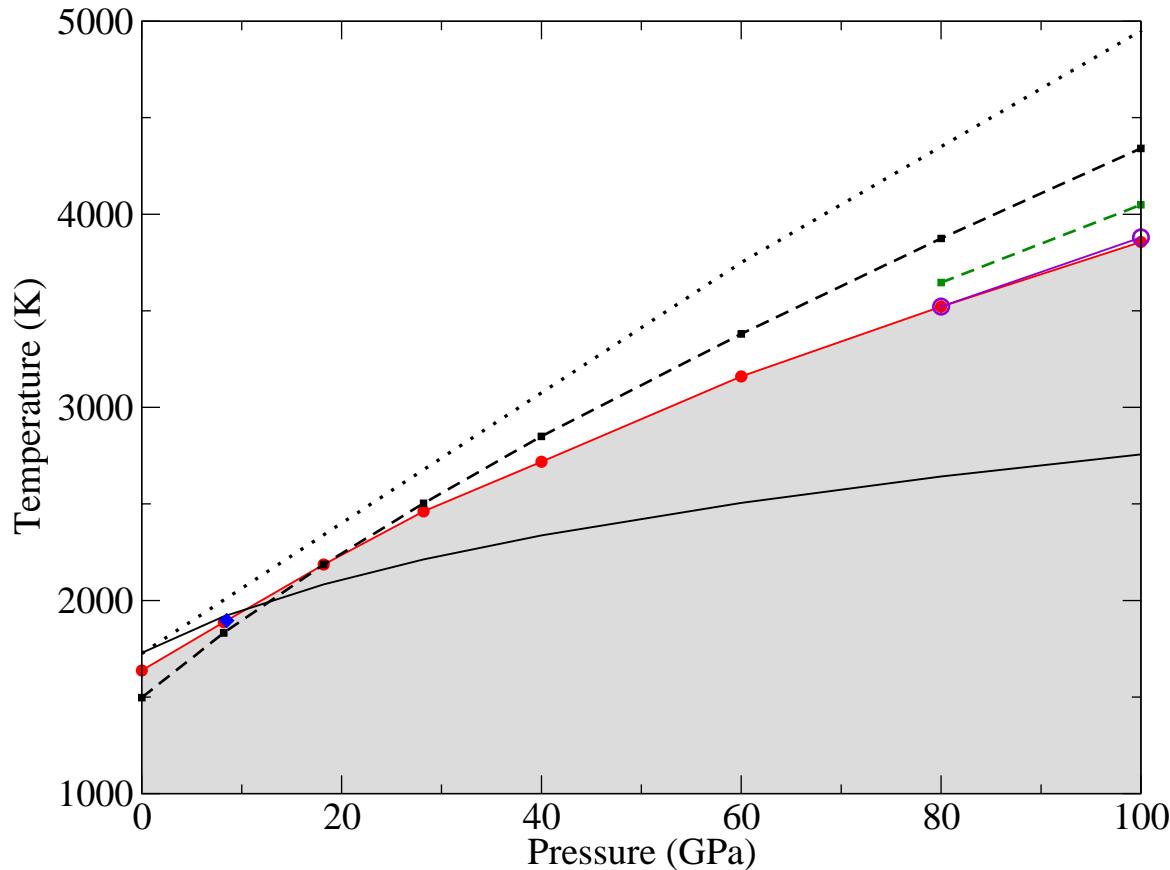
Tantalum



Molybdenum



# Melting curve of Ni



# Conclusions

- Methods for phase stability: if applied consistently give the same answer.
- Free energy
  - Small systems if reference potential is good
  - Access to thermodynamics
  - (Human) labour intensive
- Coexistence
  - Computer does most of the work
  - Large systems
  - Only melting
- Coexistence + free energy
  - Large systems only with reference potential
  - Needs good reference potentials
  - Only melting

# Acknowledgments

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  - Claudio Cazorla (ICMAB-CSIC, Barcelona)
  - Mike Gillan (UCL)
- 
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