

CRYSTALLINE and QUASI-CRYSTALLINE INTERFACES

FROM ORDER TO DISORDER

L. PRIESTER

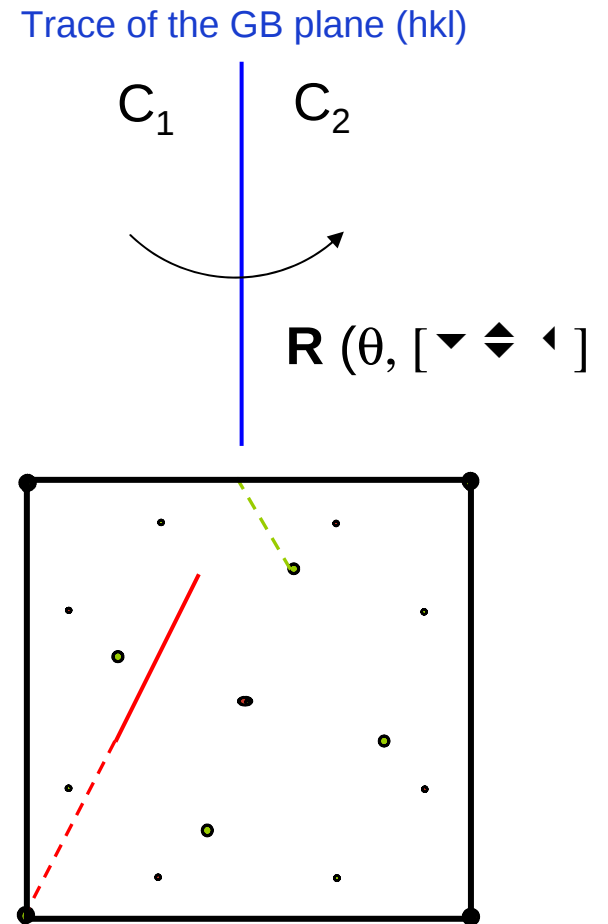
Professor emeritus Université Paris 11
ICMPE/CNRS, Thiais, France

Beijing, may 2012

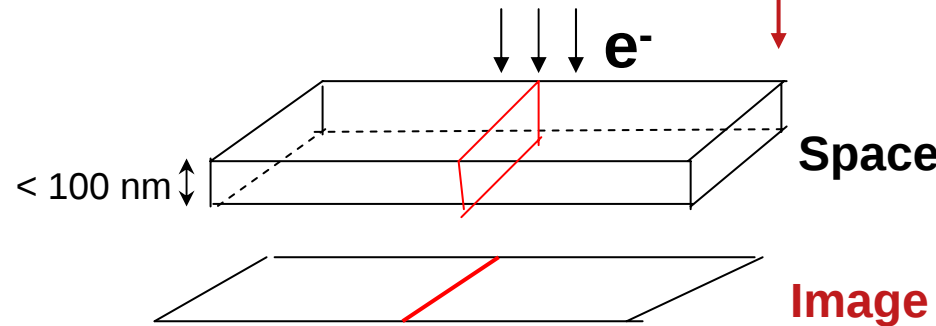
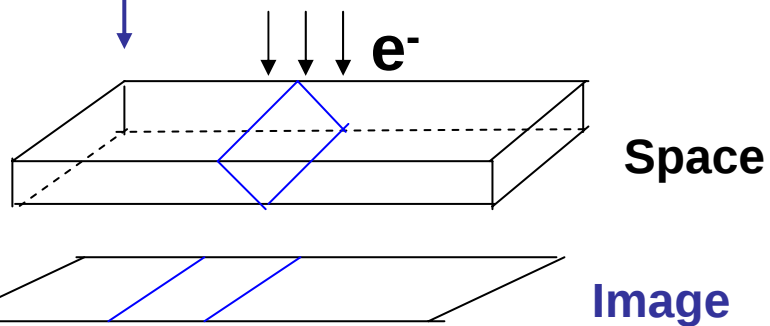
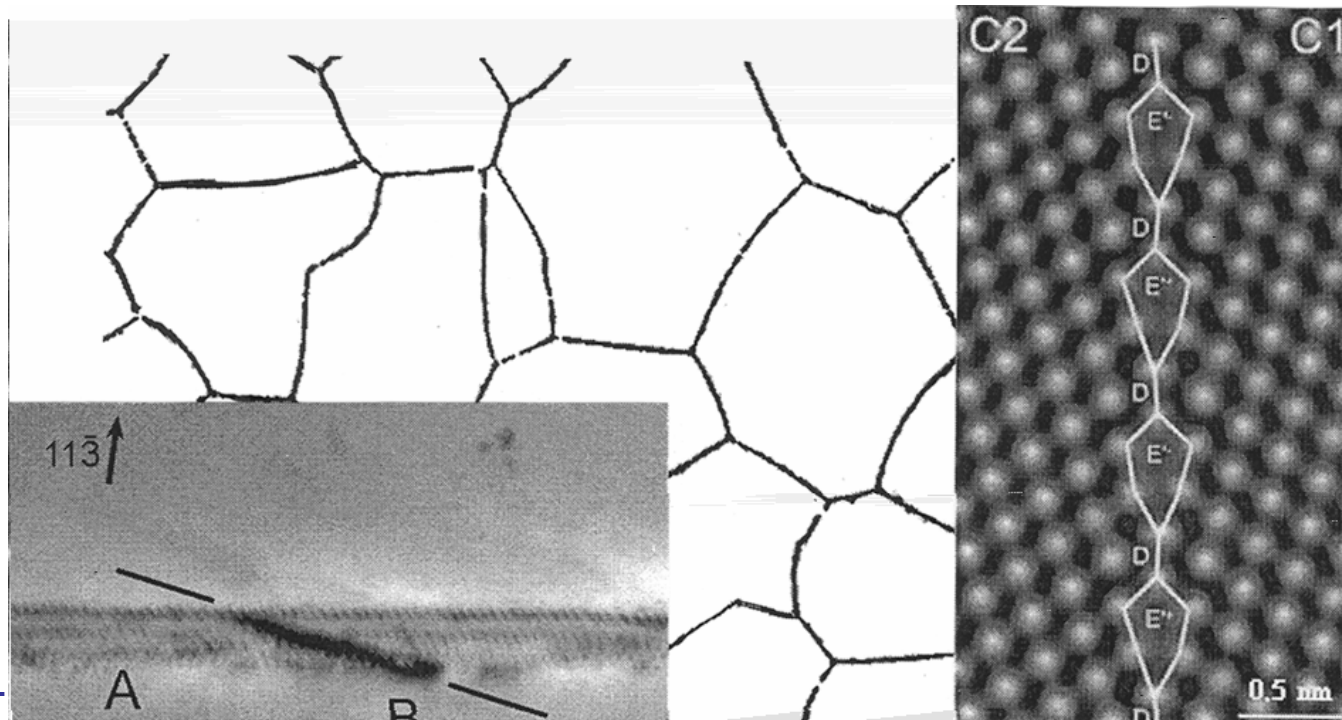
- characterized by a rotation $\mathbf{R}(\theta [uvw])$ or by a **coincidence index**

$$\Sigma = \frac{1}{\rho}$$

ρ : density of common nodes in the GB region



A GRAIN BOUNDARY at DIFFERENT SCALES



EVOLUTION OF THE CONCEPT OF GB ORDER

1 - Amorphous cement (*W. Rosenhain and D.J. Ewen, J. inst. Metals* 8 **(1912)** 149)

2 - Periodic distribution of good fit and bad fit regions

W.T. Read and W. Shockley, Phys. Rev. 78 **(1950)** 275

W. Bollmann, "Crystal defects and crystalline interfaces", Springer, Berlin **(1970)**

3 - Periodicity of structural units (SUs)

A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A309 **(1983)** 1 - 55

4 - Quasi periodicity of structural units

D. Gratias and A. Thallal, Phil. Mag. Letters, 57 **(1988)** 63

5 - Amorphous state of some GBs ?

D. Wolf, Current opinion in Solid State and Materials Science 5 **(2001)** 435.

} Outline

EVOLUTION OF THE CONCEPT OF GB ORDER

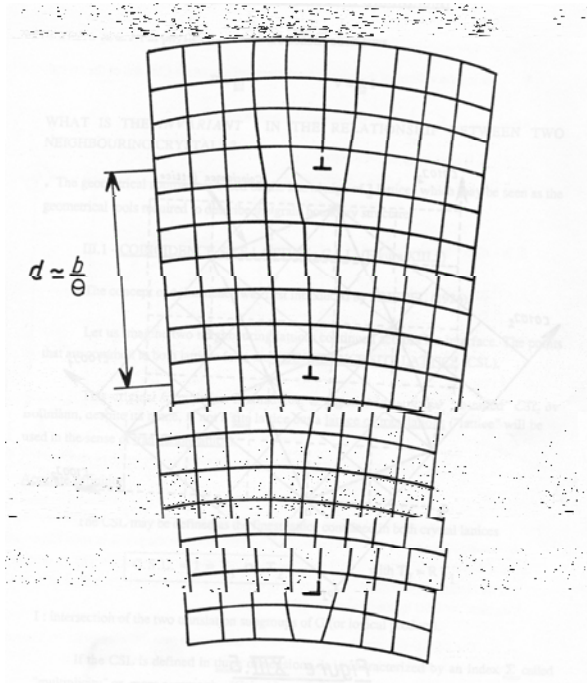
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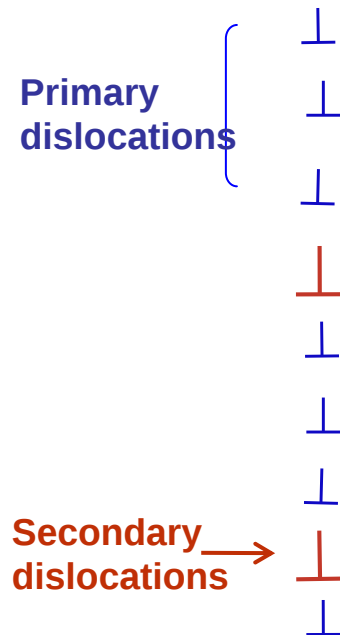
W.T. Read and W. Shockley, *Phys. Rev.* 78 (1950) 275

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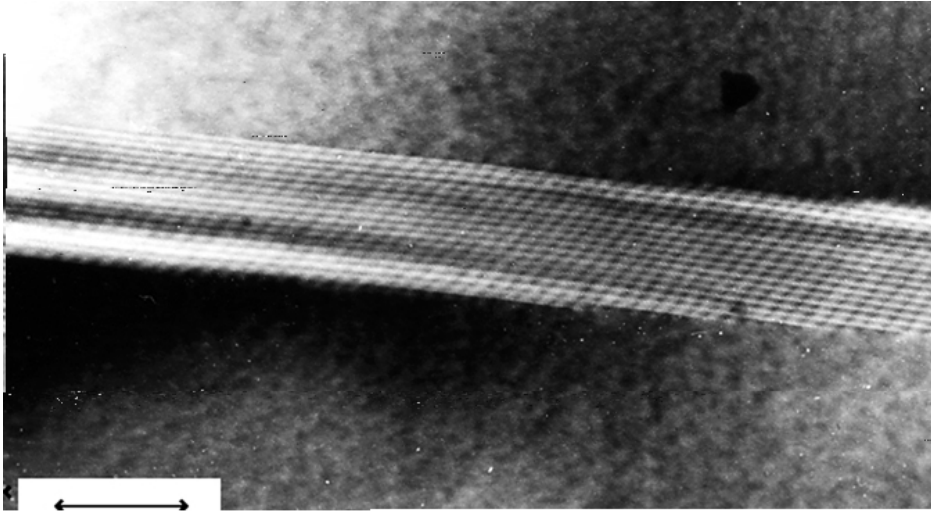
For low angle tilt GB



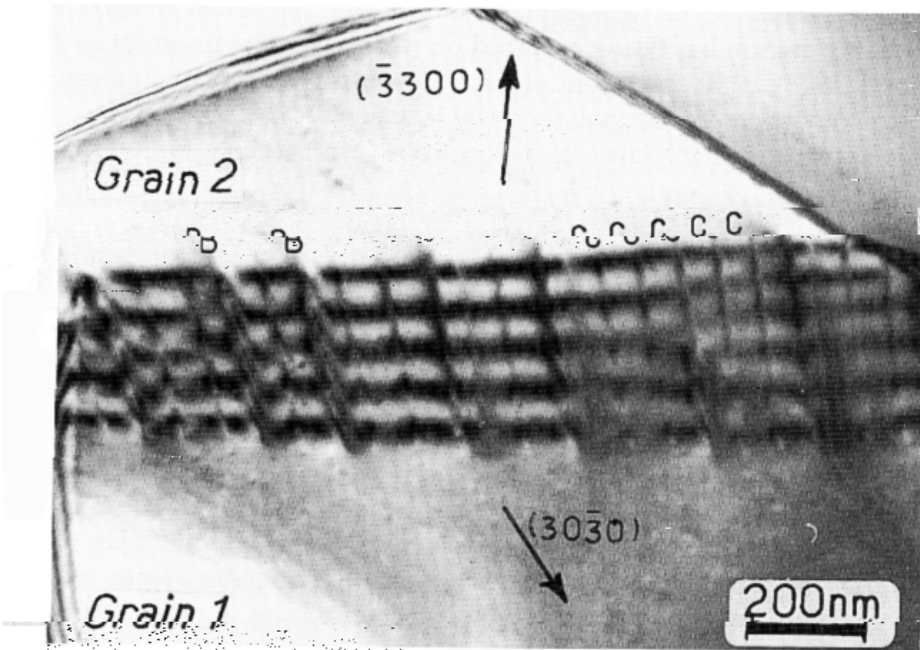
For any GB : intrinsic dislocations



Some examples of intrinsic dislocations



Primary intrinsic dislocations in low - angle (2°) grain boundary in a Fe-Mo alloy



Secondary intrinsic dislocations in a high-angle (85.5°) grain boundary in alumina (oxide)

OUTLINE

The structural unit model

Periodicity of structural units (SUs)

A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A309 (1983) 1 - 55

Quasi-crystalline interfaces

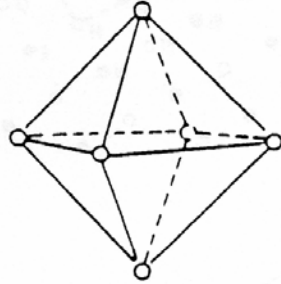
Quasi periodicity of structural units

D. Gratias and A. Thallal, Phil. Mag. Letters, 57 (1988) 63

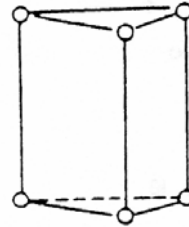
Amorphous state of some GBs ?

D. Wolf, Current opinion in Solid State and Materials Science 5 (2001) 435.

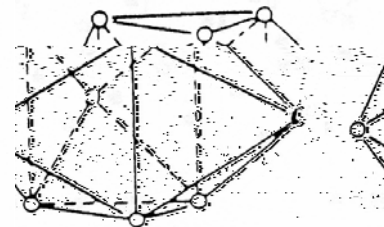
STRUCTURAL UNIT $\equiv$ POLYHEDRAL CLUSTER OF ATOMS



Tetrahedron

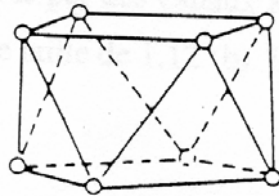


Octahedron

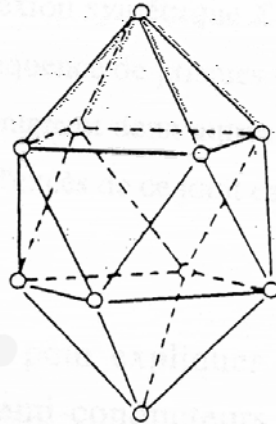


Trigonal prism

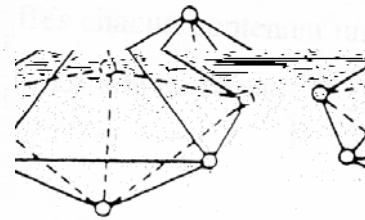
Capped trigonal prism



Archimedean square antiprism



Capped Archimedean square antiprism



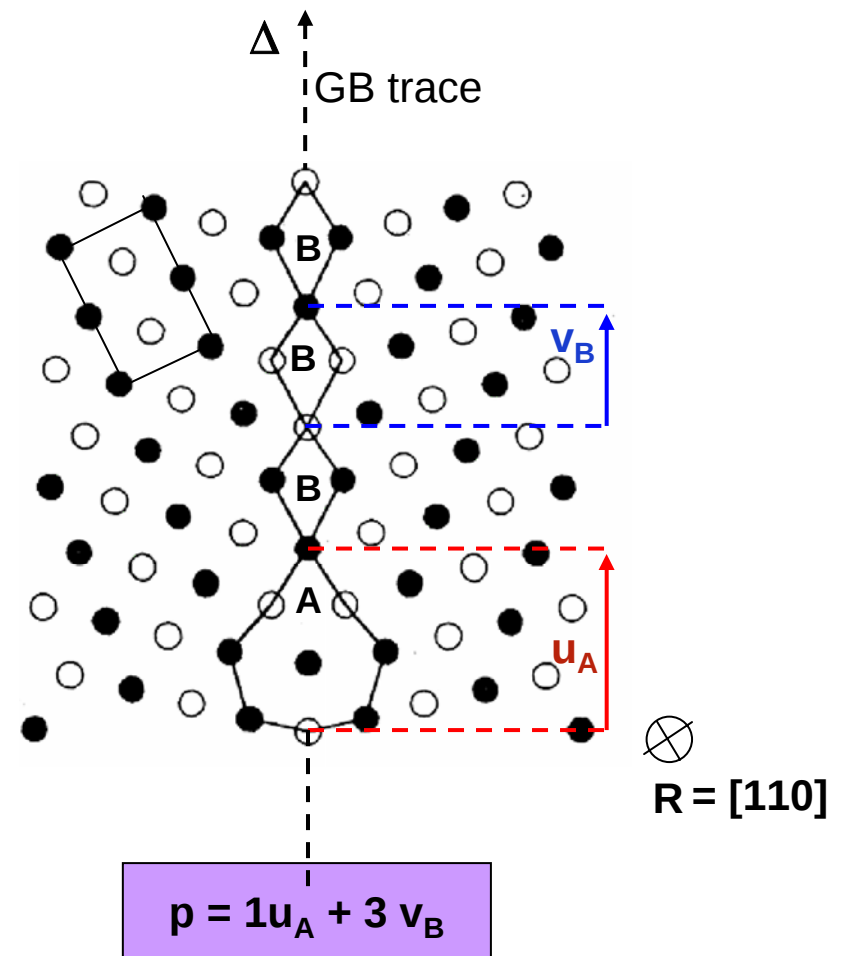
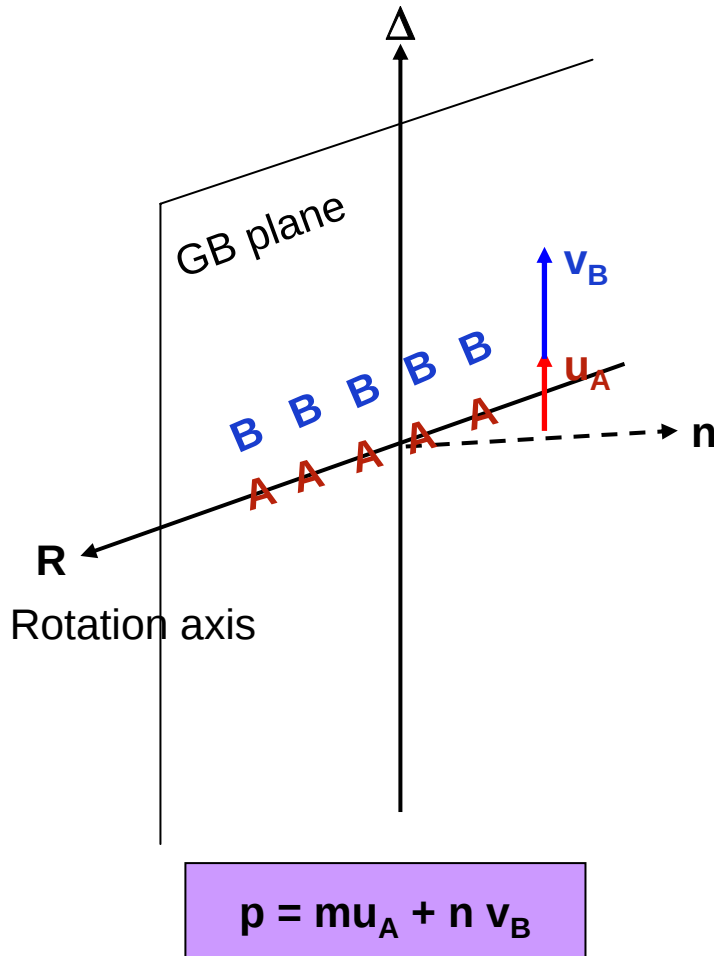
Pentagonal bipyramid

Equivalent to the elemental cells in crystals (cube, hexagon ...)

Limited number of polyhedra

Analogy with the hard sphere model of liquid structure - 5 similar clusters (*Bernal - 1964*)

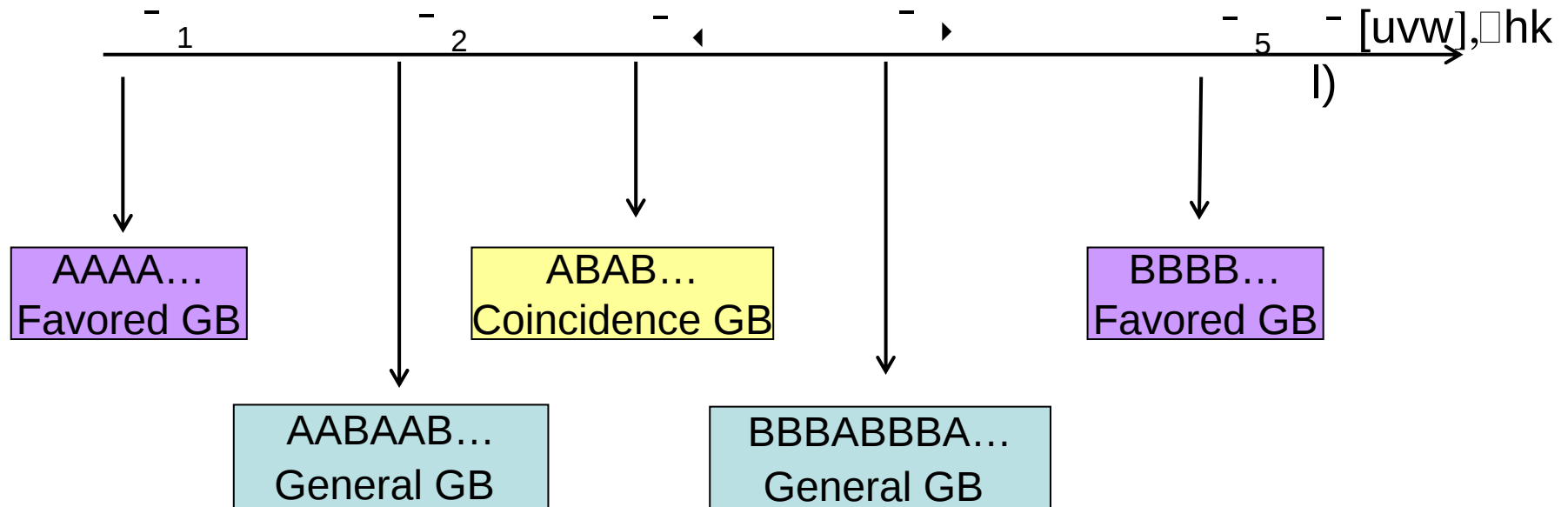
STRUCTURAL UNIT MODEL GEOMETRY



- Rational ratio $m/n \Rightarrow$ Periodic grain boundary
- Irrational ratio $m/n \Rightarrow$ Quasi-periodic grain boundary

STRUCTURAL UNIT MODEL PRINCIPLE

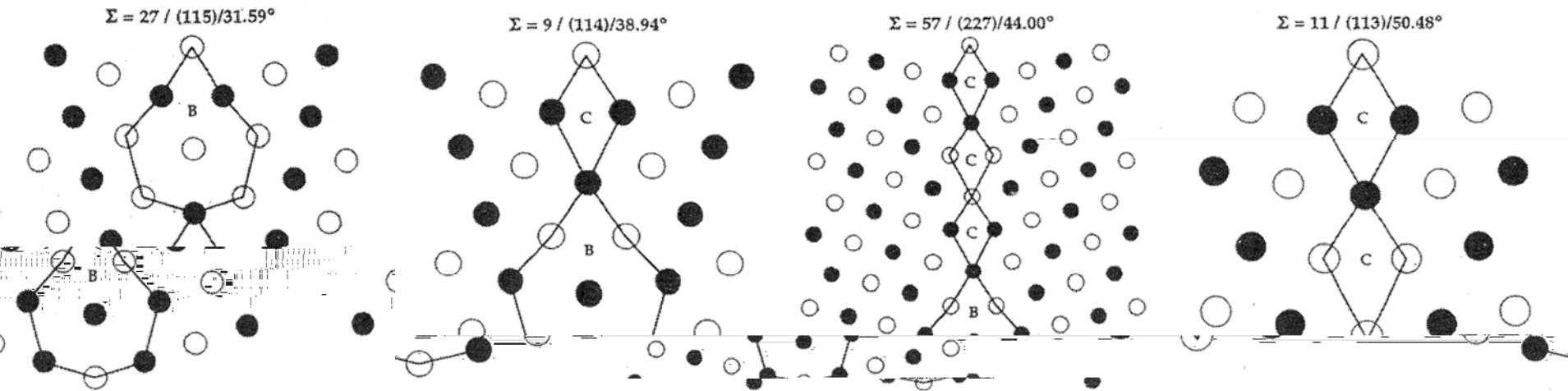
Any long period GB may be described as a sequence of structural units of two short period (favored) GBs.



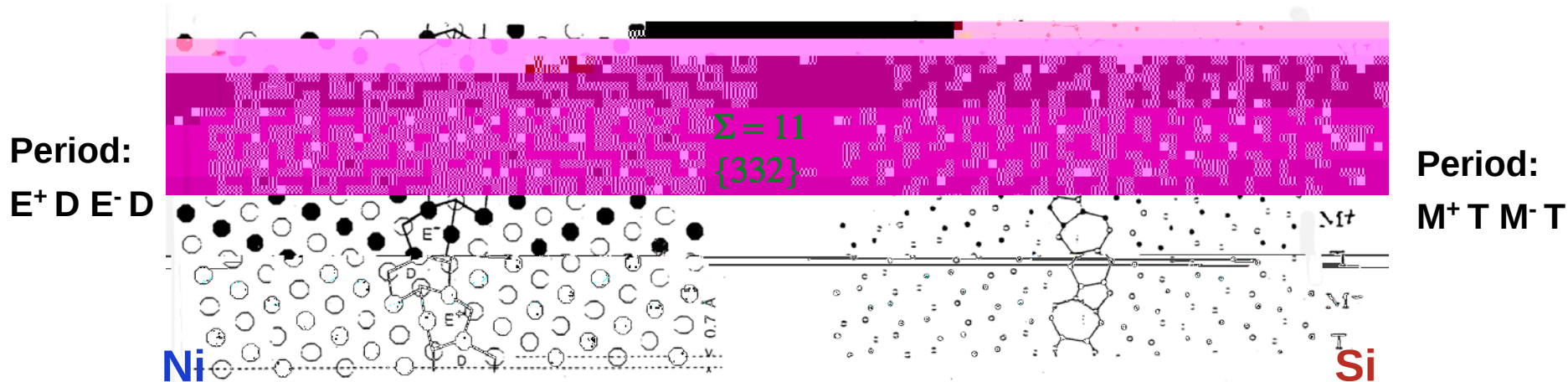
simple principle **BUT COMPLEXITY**

- Multiplicity of descriptions
- Distortion of the SUs
- Hierarchy of descriptions

Series for symmetrical tilt GB around $\langle 110 \rangle$ for aluminium (FCC)

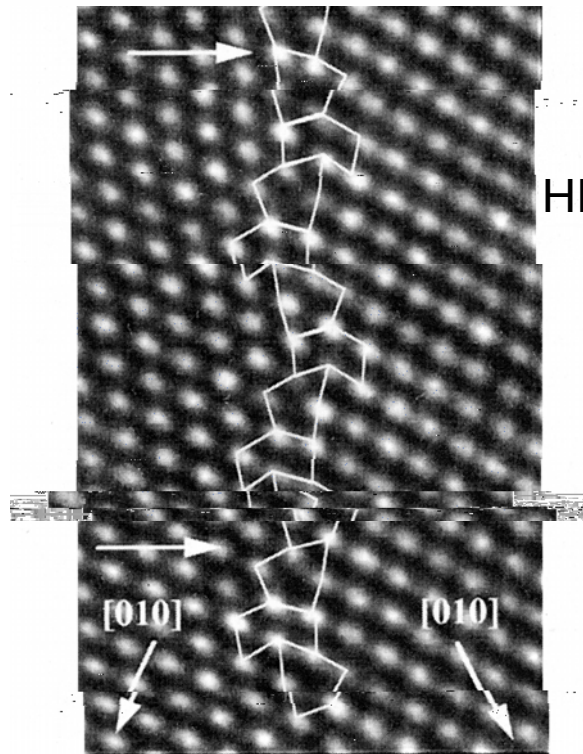
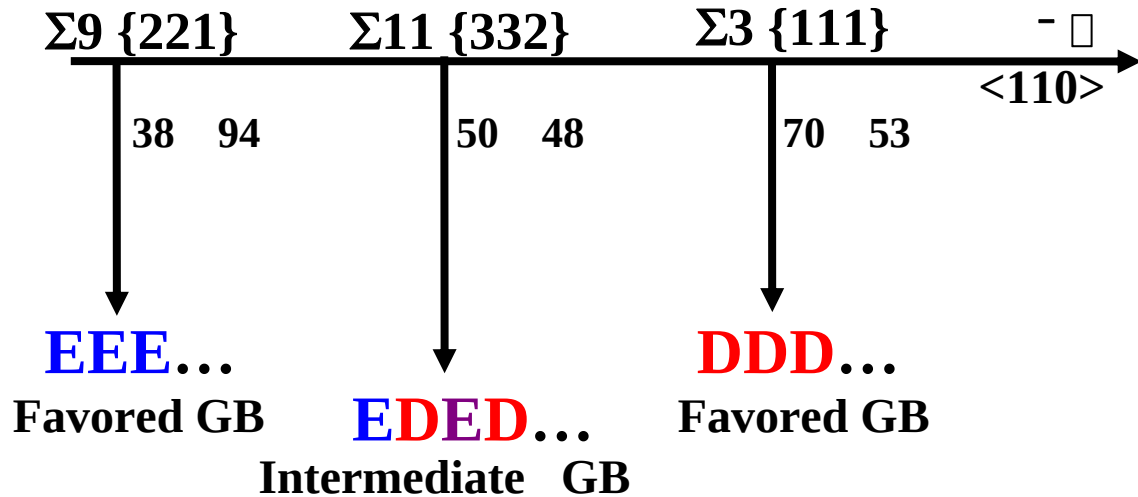
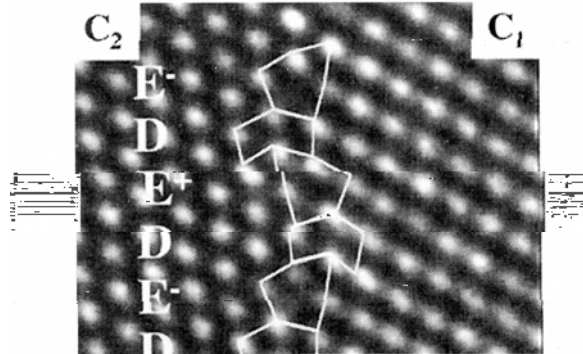


A given GB (same R and θ) in different materials



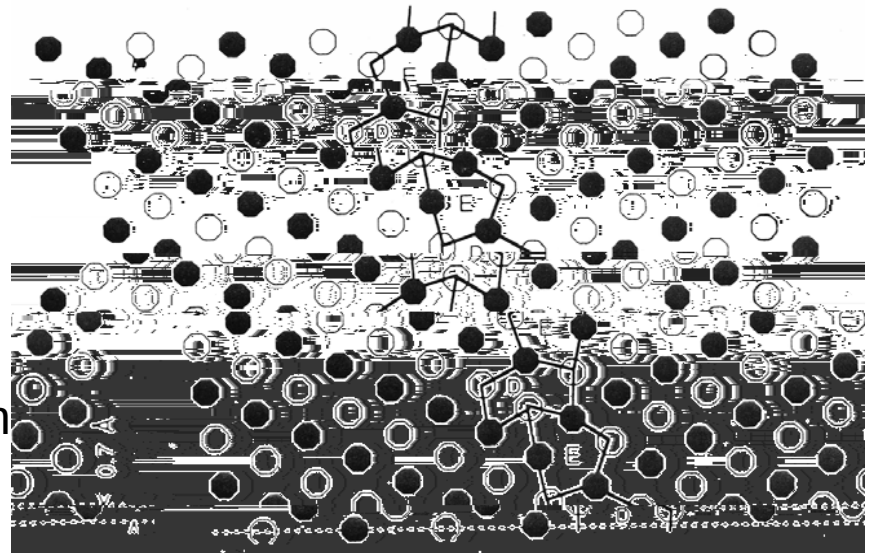
The shapes of the structural unit differs but the period is similar

Description in terms of Structural Units (SU)



HRTEM image

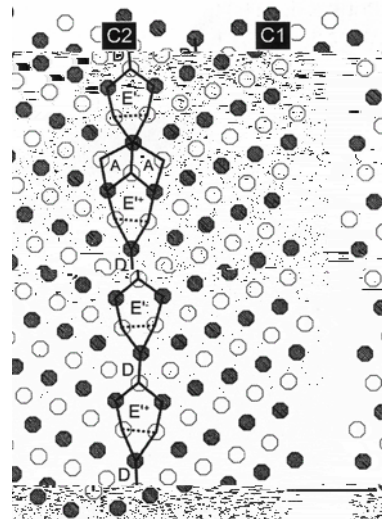
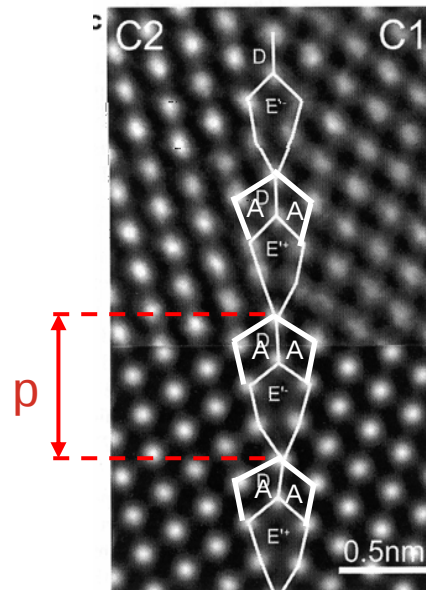
Simulation



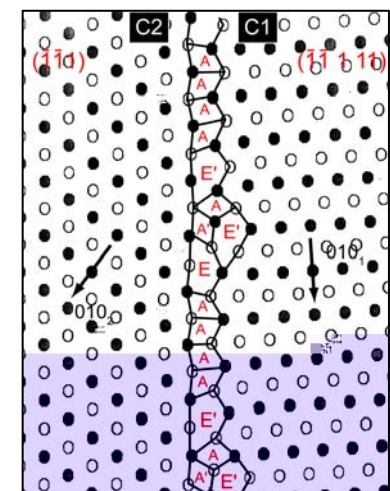
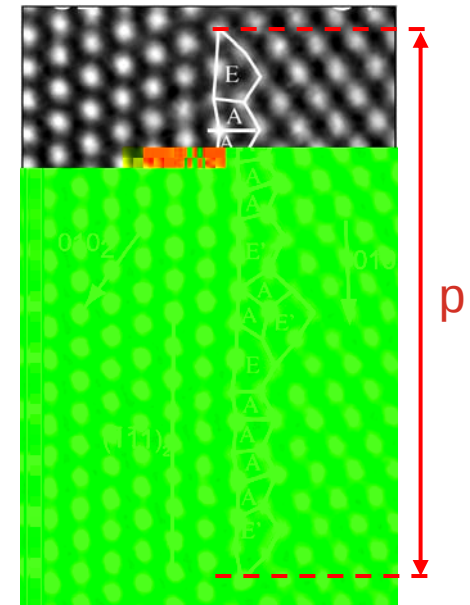
VALIDITY of SU MODEL for FACETTED GB - Near $\square 9$ (Cu)



Symmetrical $\{221\}$ facet



Asymmetrical facet

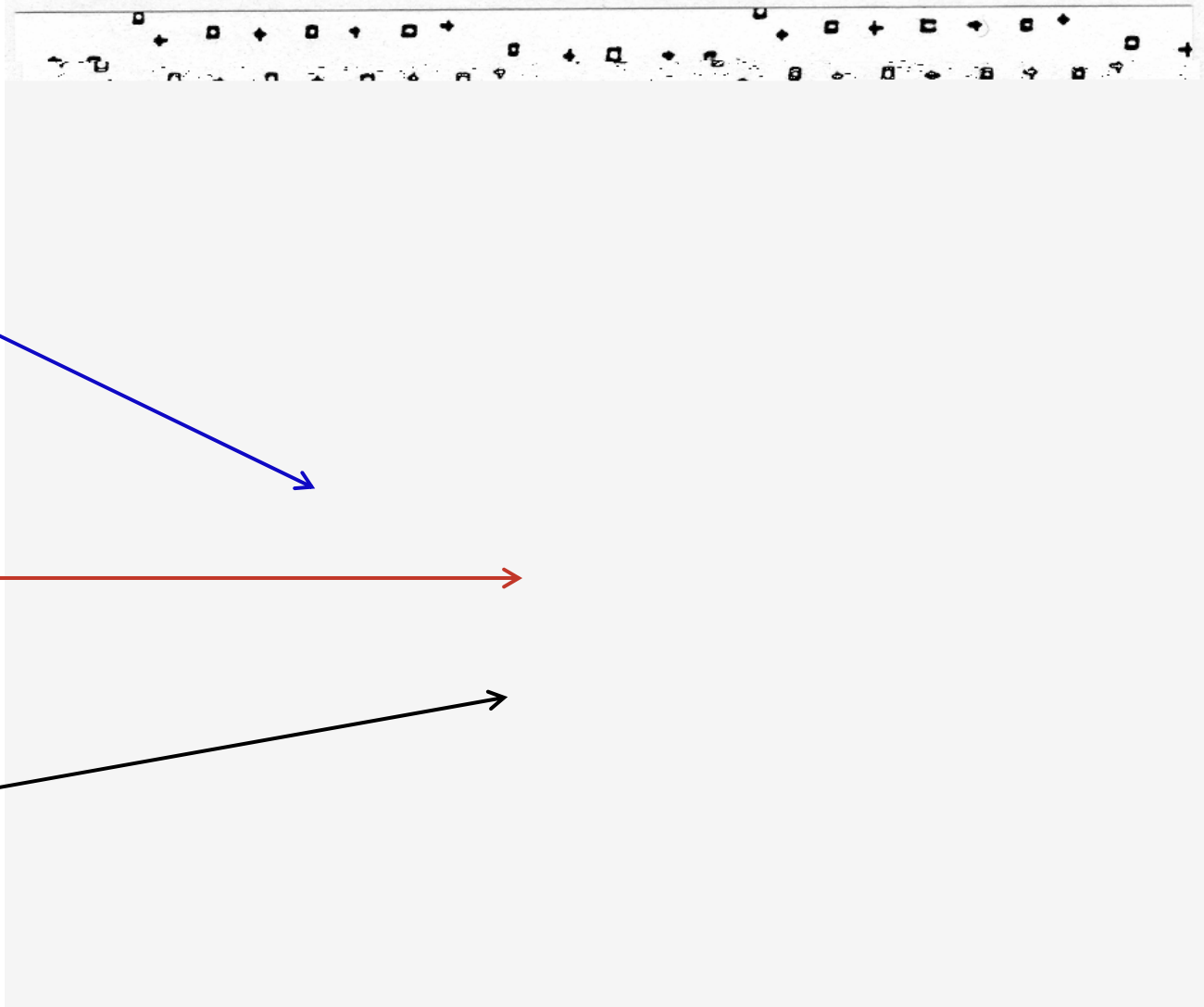


STRUCTURAL UNIT MODEL FOR TWIST GBs

Example of $\Sigma 85$ - 8.80 [001]

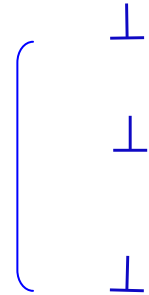
B Unit

Filler



“STRUCTURAL UNITS/ INTRINSIC DISLOCATIONS

Primary
Dislocations ($\perp$)
 $b = b_m$



A

A

A

Majority units

B

← Minority unit

A

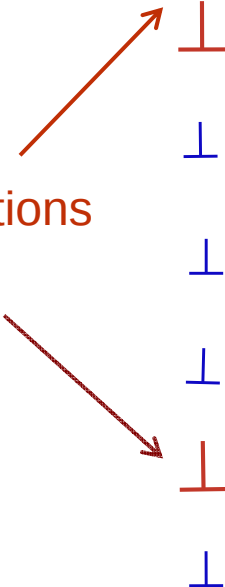
A

A

B

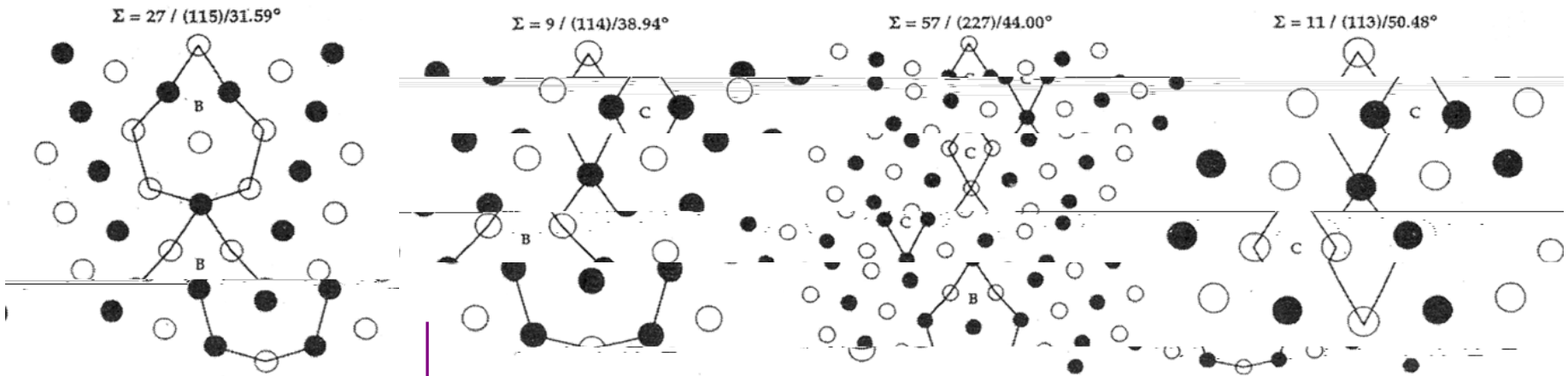
A

Secondary Dislocations
($\perp$ -)
 b_{DSC}



“Minority units/secondary Dislocations” disturb the periodicity of “majority units /primary dislocations”

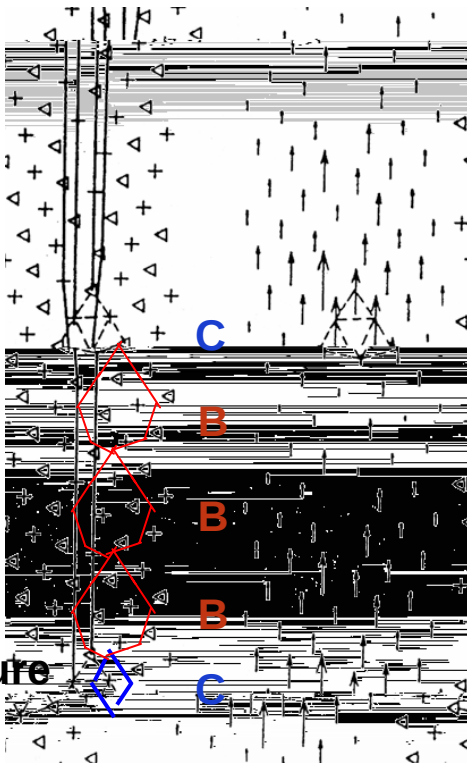
HRTEM IMAGES and HYDROSTATIC STRESS FIELDS



$\Sigma = 89 (229)$

Predicted structure

θ	Σ	Plan du joint	Structure
31.59	27	(115)	B.B
34.89	89	(229)	□□□C
38.94	9	(114)	□C
50.48	27	(113)	C.C



Hydrostatic stress fields

Simulated structure

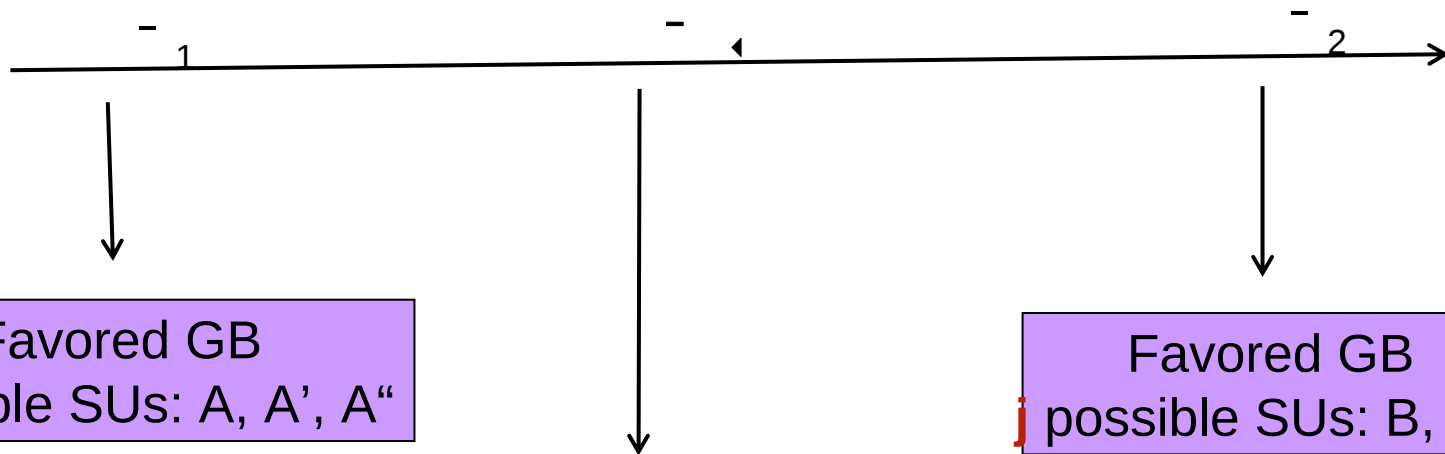
BBBC

STRUCTURAL UNIT MODEL : Multiplicity of descriptions

A favored GB may be described by different SUs whose the energies are very



Any intermediate GBs may be constituted by different combinations N of these



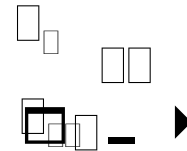
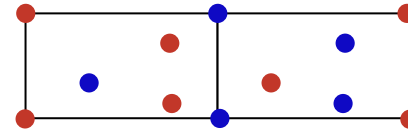
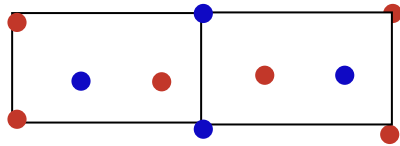
GB period : $\mathbf{p} = m \mathbf{u}_A + n \mathbf{v}_B$

$$N = i^m j^n$$

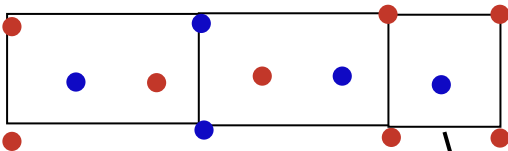
All the N configurations are not stable
 Comparisons with the hydrostatic stress field and with the HRTEM images

Examples of multiplicity of descriptions

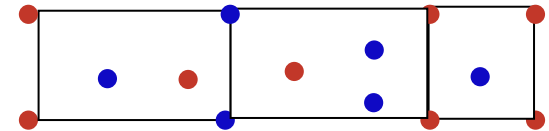
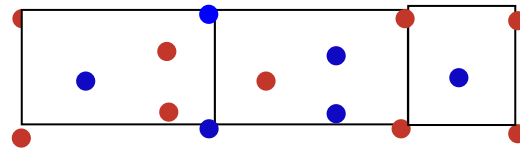
Favored tilt GB $\square 5$ (210) - 36.9 $[001]$



Coincidence GB $\square 17$ (530) - 28.1 $[001]$



A: perfect crystal

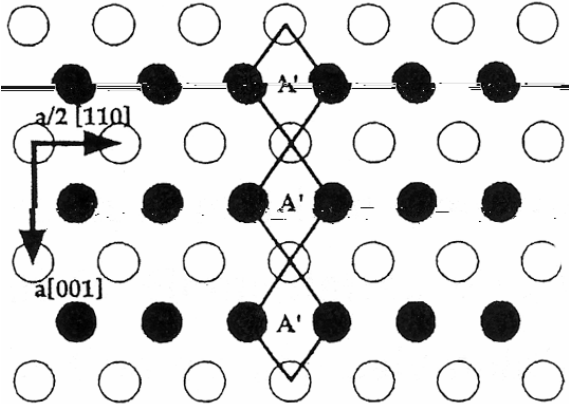


$$N = 2^2 \cdot 1^1 = 4$$

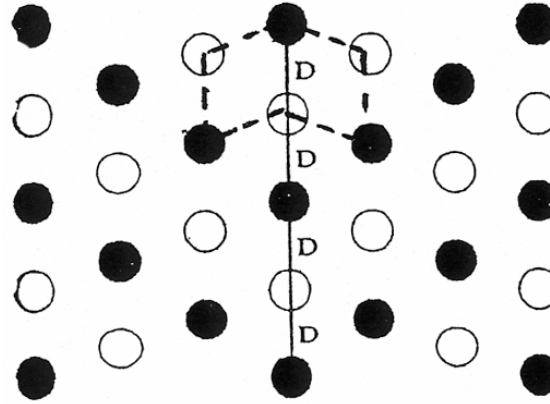
Energy ratio: 1.07 / 1.09/

STRUCTURAL UNIT DISTORTION

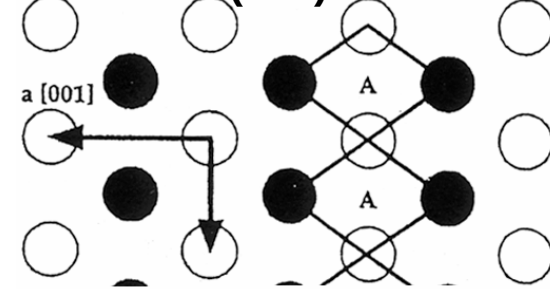
$\Sigma = 1 (110) 0$



$\Sigma = 3 (111) 70.5$



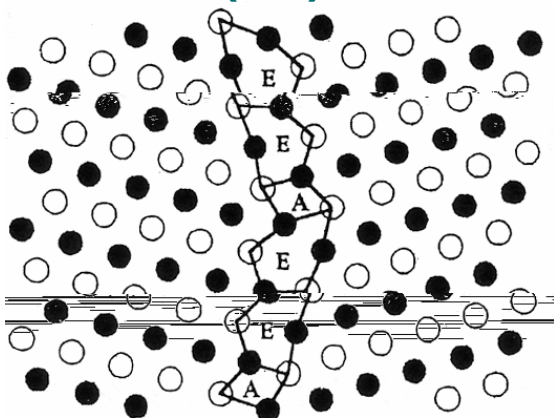
$\Sigma = 1 (001) 180$



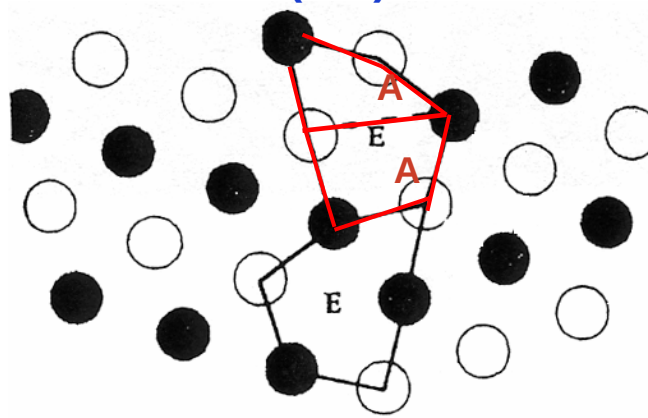
$\Sigma = 1$ (single crystal): same unit A and A' rotated by 180°

$\Sigma = 3$ (twin): unit D $\equiv$ 2 A units rotated by 70.5°

$\Sigma = 27 (552) 32.5$



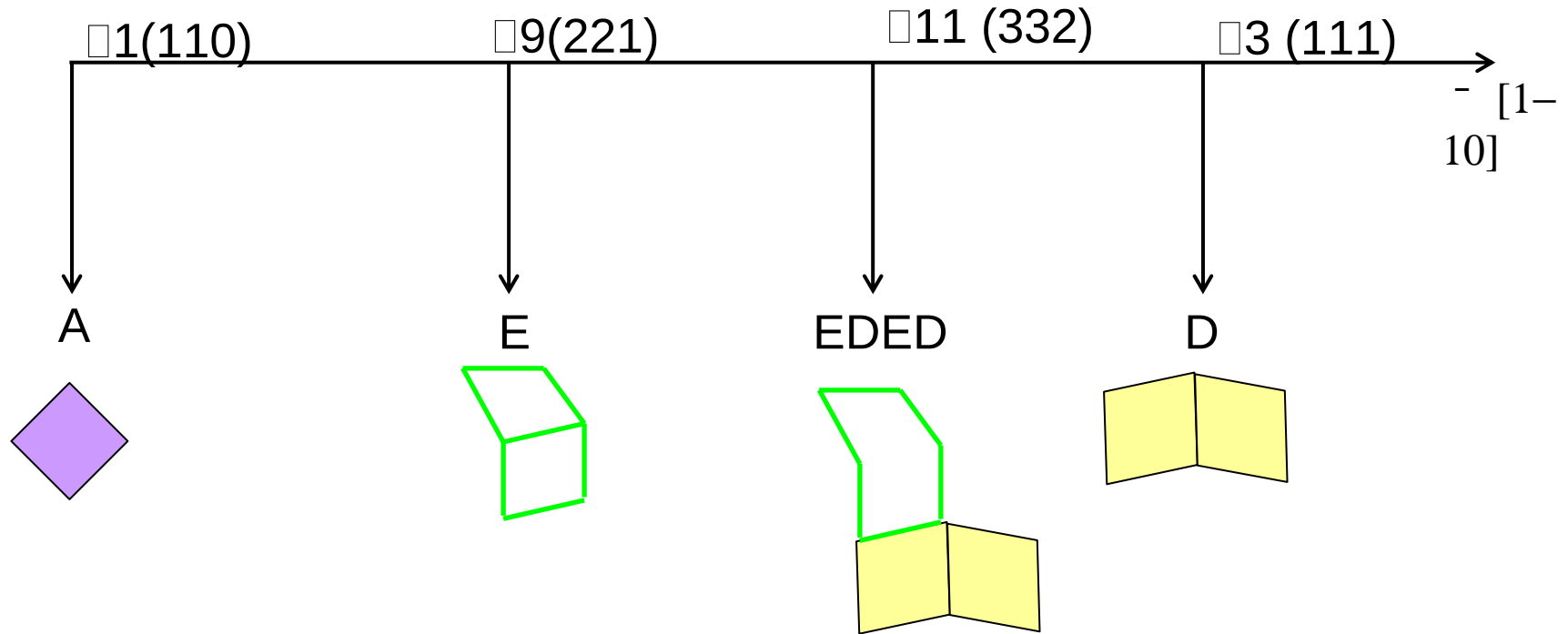
$\Sigma = 9 (221) 38.9$



$\Sigma = 9$: unit E formed by two distorted and rotated A units

$\Sigma = 27$: period = EEA but some E units are distorted

SU DISTORTION $\Rightarrow$ HIERARCHY of GB DESCRIPTIONS



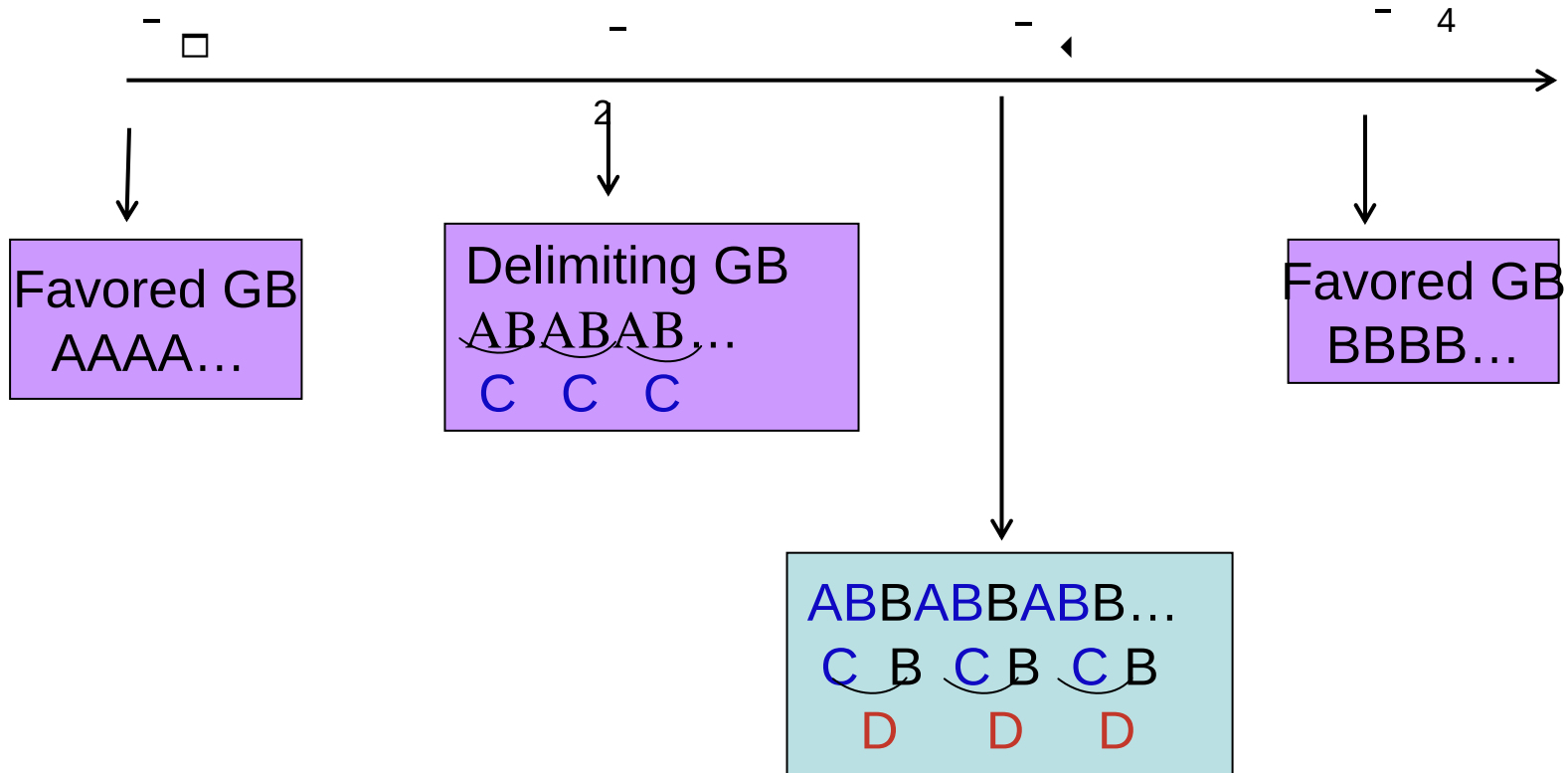
$\square 9(221)$ could be described by A and D units but strong **distortion**



Better description by E unit $\Rightarrow$ then use of E for the structure of $\square 11(332)$

$\square 9$ appears as a **delimiting GB**

HIERARCHY OF GB DESCRIPTIONS



General rational GBs
(rational ratio m/n of A and B units)

- As the order of the description increases $\Rightarrow$ the distortions of the SUs decrease
- The atomic description requires the knowledge of the basic structures

HOW TO GENERATE the SEQUENCE of SUs ?

$$\mathbf{p} = m\mathbf{u}_A + n\mathbf{v}_B$$

There is a huge number of ways for arranging m units A and n units B in a periodic fashion

$$W = \frac{(m + n - 1)!}{m! n!} \quad (\text{For } m = 13 \text{ and } n = 19, W = 10.855.425)$$

THUS

To determine the sequence of structural units, it is necessary to use:

- **an algorithm**

A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A 309 (1983) 1.

Main assumption: The boundary structure changes in as smooth and continuous manner as possible when θ varies

- **a strip band method** (analogous to what is used for quasicrystallography),

A.P. Sutton, Prog. Mat. Sci. 36 (1992) 167.

ALGORITHM to DETERMINE THE S.U. SEQUENCE in a GB

Principle - GB period = $mA + nB$ with $(m < n)$

Ex: $\Sigma = 4233$ (16,16,61) - period = $13A + 19B$

$$\frac{1}{p+1} < \frac{m}{n} < \frac{1}{p}$$

$$\underbrace{A(p+1)B}_{\text{C}} \Leftarrow X \Rightarrow \underbrace{ApB}_{\text{D}}$$

$$r \text{ C} + s \text{ D} \rightarrow$$

$$\begin{aligned} r &= n - mp \\ s &= m(p+1) - n \end{aligned}$$

$$\frac{1}{2} < \frac{13}{19} < \frac{1}{1}$$

$$\underbrace{ABB}_{\text{C}} \Leftarrow X \Rightarrow \underbrace{AB}_{\text{D}}$$

$$\begin{aligned} r \text{ C} + s \text{ D} \\ r = 6 ; s = 7 \end{aligned}$$

$$\frac{1}{q+1} < \frac{r}{s} < \frac{1}{q}$$

$$\underbrace{C(q+1)D}_{\text{E}} \Leftarrow X \Rightarrow \underbrace{CqD}_{\text{F}}$$

$$t \text{ E} + v \text{ F}$$

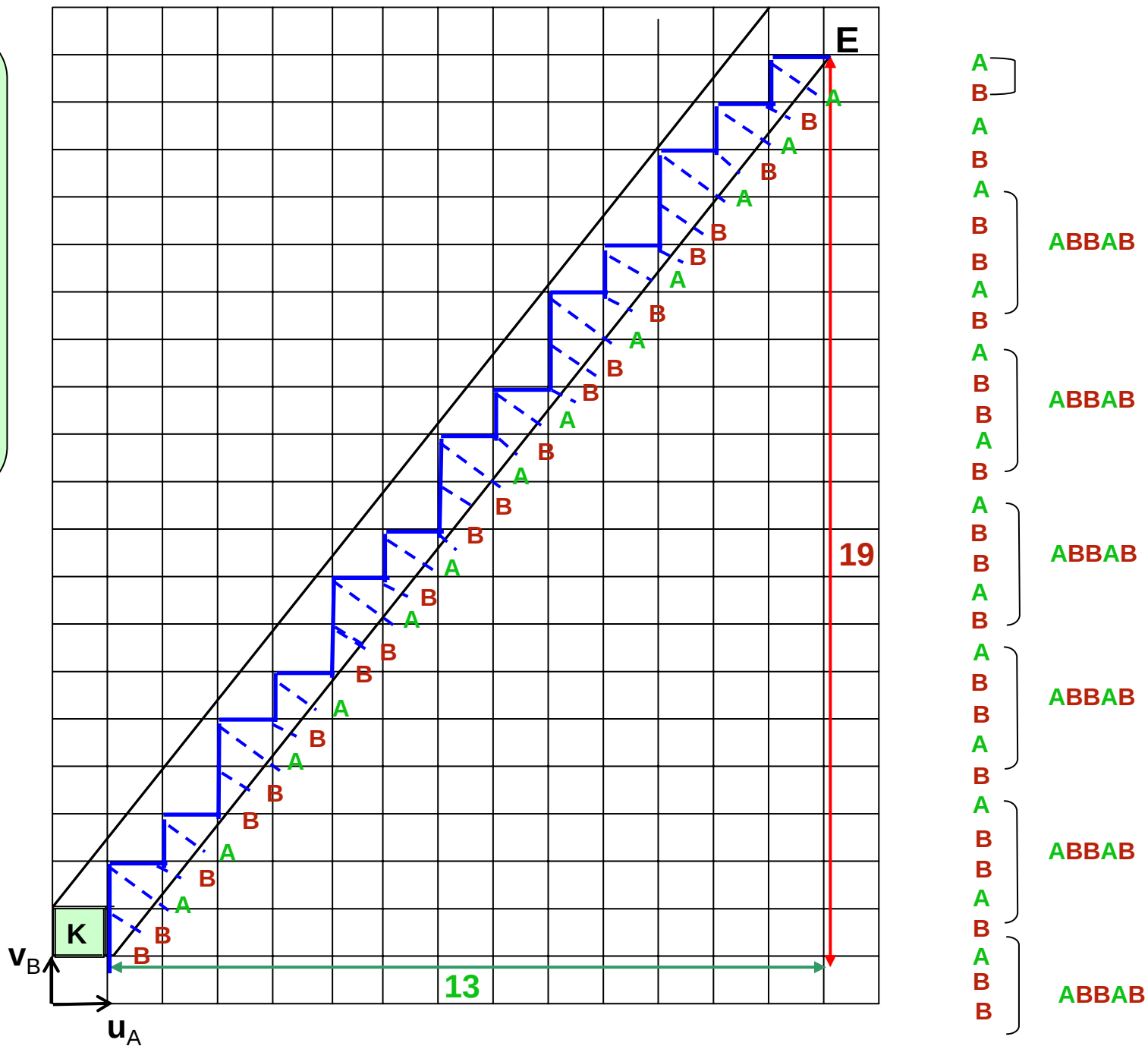
$$\frac{1}{2} < \frac{6}{7} < \frac{1}{1}$$

$$\underbrace{CDD}_{\text{E}} \Leftarrow X \Rightarrow \underbrace{CD}_{\text{F}}$$

$$1 \text{ E} + 5 \text{ F}$$

and so until the number of minority units = 1

**STRIP
METHOD
for
determining
the SU
sequence
in a GB**



OUTLINE

The structural unit model

Periodicity of structural units (SUs)

A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A309 (1983) 1 - 55

Quasi-crystalline interfaces

Quasi periodicity of structural units

D. Gratias and A. Thallal, Phil. Mag. Letters, 57 (1988) 63

Amorphous state of some GBs ?

D. Wolf, Current opinion in Solid State and Materials Science 5 (2001) 435.

HOW TO GENERATE QUASIPERIODIC SEQUENCES

ALGORITHM *(Levine and Steinhard, 1984)*

For irrational tilt GBs: $m_A / n_B = \underbrace{m/n}_{\text{rational}} + \underbrace{\lambda}_{\text{irrational}}$

More simple quadratic form such as:
 $\lambda^2 - \lambda - 1 = 0$ in that case
 $\lambda_1 = \tau = (1 + \sqrt{5})/2$

Golden number

$$\mathbf{u}_A = \begin{bmatrix} 1 \\ 0 \end{bmatrix} \quad \mathbf{v}_B = \begin{bmatrix} 0 \\ 1 \end{bmatrix}$$

Self-similar sequence obtained by applying the operation $M = \begin{bmatrix} 1 & 1 \\ 1 & 0 \end{bmatrix}$
(det M = -1)

Then repeat...

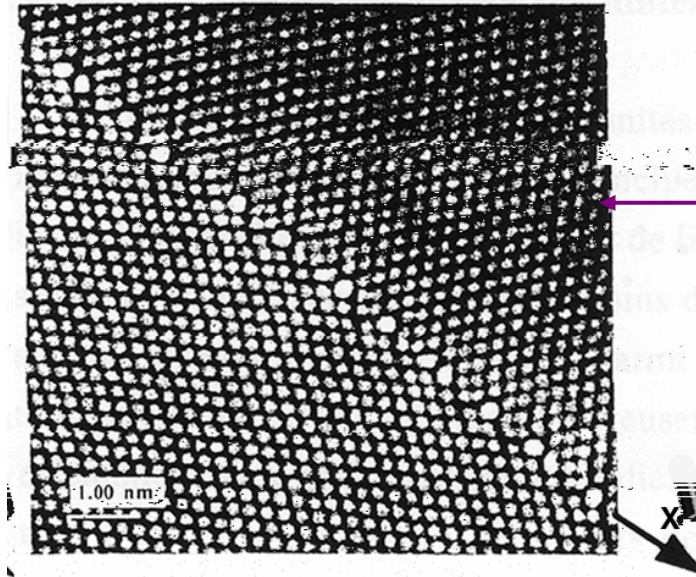
Number of iterations	Sequence of US	m_A / n_B
0	AB	1/1
1	BAB	1/2
2	BABBA	2/3
3	BABBABAB	3/5
4	BABBABABBABBA	5/8
↓	↓	↓
∞	Quasi-periodicity	1/τ

Quasiperiodic GBs are the limits of periodic GBs with increasing periods

STRIP METHOD

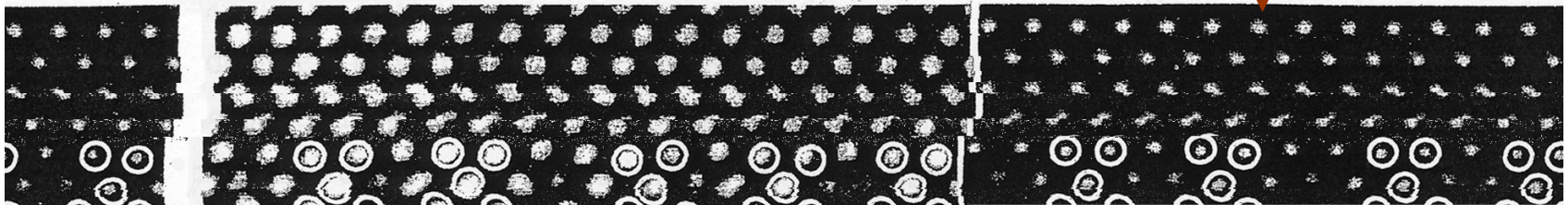
Irrational slope of the E line in the section/projection method

Quasiperiodic Structure of a GB in gold



□R□□□ image of a quasi periodic GB
90 [001] (100)_I // (110)_{II}

Simulated image based on calculation
of the GB structure
(approximant 29/41 $\varepsilon = 3 \cdot 10^{-4}$)



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Quasi-crystalline interfaces

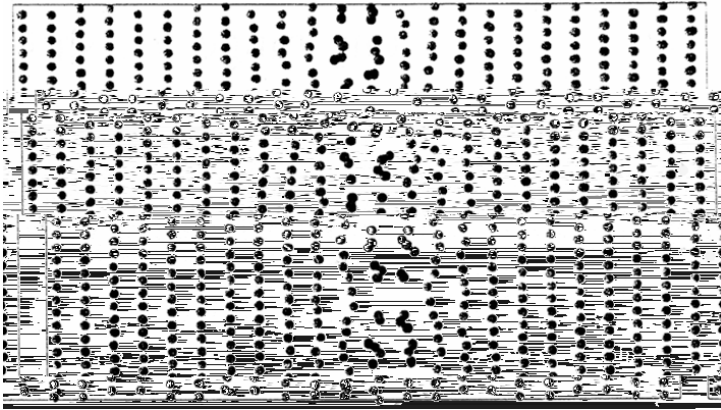
Quasi periodicity of structural units

D. Gratias and A. Thallal, Phil. Mag. Letters, 57 (1988) 63

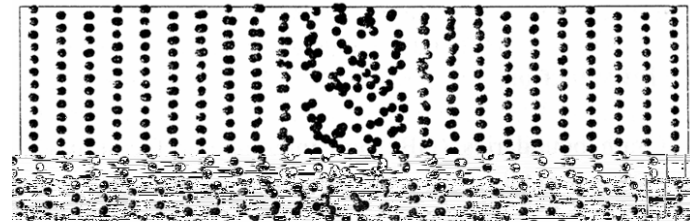
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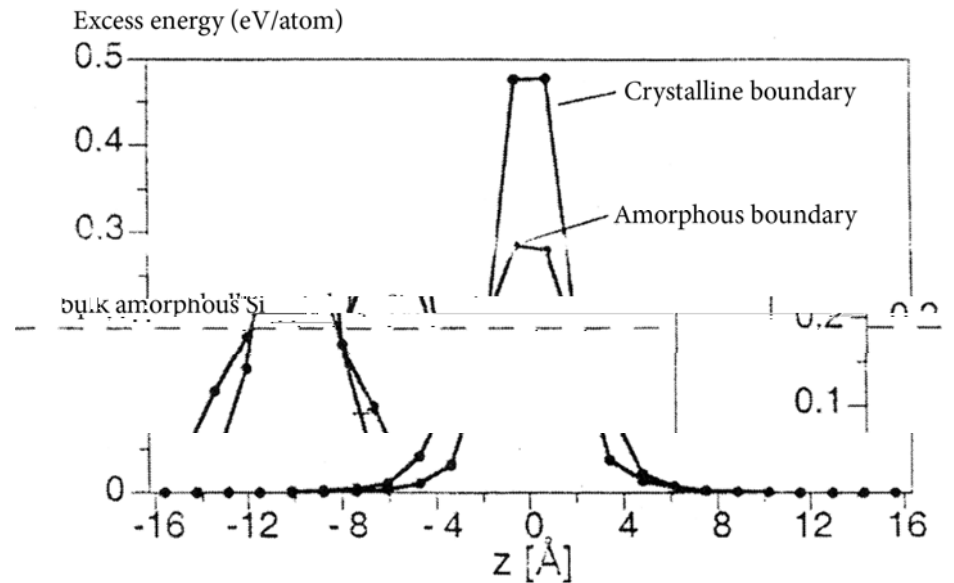
CRYSTALLINE / AMORPHOUS STATE of a $\Sigma = 29$ TWIST GB in SILICON



Crystalline GB (relaxed at low T)



Amorphous GB (relaxed at high T)



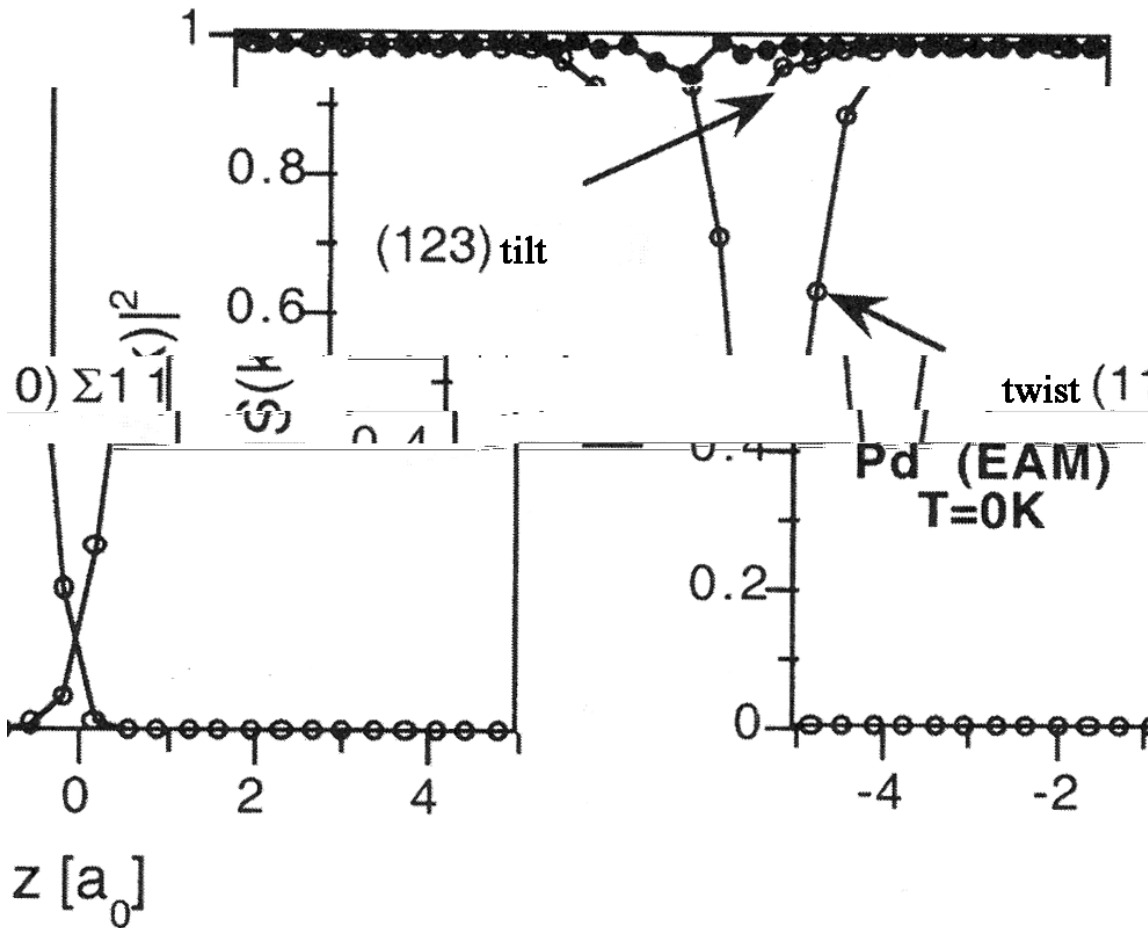
(c)

GB ORDER / DISORDER ?

Distinction between ORDER and ENERGY

ENERGY

- not controlled by the order at large distances (periodicity))
- controlled by the short-distance order or local arrangement of atoms (



The square of the structure factor $S(k)^2$ is function of the crystallinity
 $= 1$ (if 100% crystal)

Tilt GB is crystalline

Twist GB is amorphous

Although

$$E_{(110) \text{ twist}} < E_{(123) \text{ twist}}$$

